






Review

# Corrosion Behaviour in CO<sub>2</sub> Pipeline Transport: A Review of the Impact of Condensates and Impurities

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## Highlights

- Role of the principal CCTS strategy and effect on the resulting flux
- The role of water in the CO<sub>2</sub> flux and Top of the line corrosion
- Role of the different impurities on the corrosion behaviour

## Abstract

The high emissions of carbon dioxide (CO<sub>2</sub>) into the atmosphere have driven the development of carbon capture, transport, and storage (CCTS) technologies. These focus on capturing CO<sub>2</sub> from industrial exhaust gases and transporting it through existing pipeline networks. Although various capture techniques are available, they may introduce impurities such as O<sub>2</sub>, N<sub>2</sub>, Ar, H<sub>2</sub>O, NH<sub>3</sub>, and others into the CO<sub>2</sub> stream. These contaminants can significantly alter the thermophysical behaviour of the fluid, making the phase behaviour predictions, reliable for pure CO<sub>2</sub>, much more complex. Pressure and temperature variations along pipelines can induce unexpected phase transitions, affecting fluid composition and potentially triggering corrosion. This review examines the formation of condensates within pipelines and their role in initiating corrosion phenomena, with a focus on top of the line corrosion (TLC) and conventional CO<sub>2</sub>-induced corrosion (sweet corrosion). The main literature findings highlight how phase changes and altered fluid composition due to corrosion processes can significantly intensify degradation mechanisms during CO<sub>2</sub> transport.

**Keywords:** energy transition; CCTS; impurities; corrosion rate; top of the line corrosion



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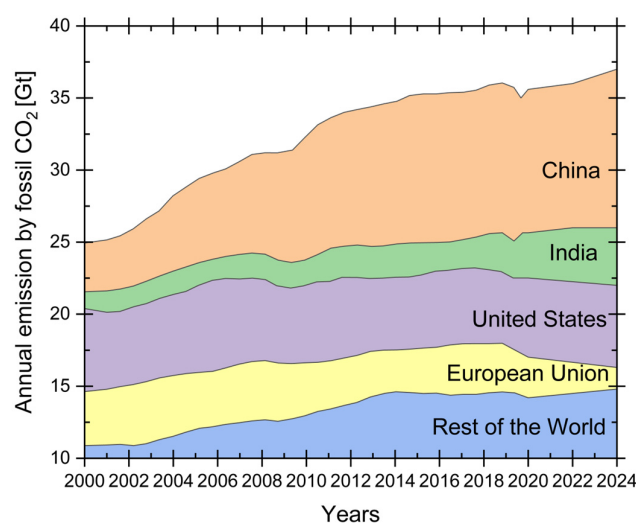
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## 1. Introduction

### 1.1. Energy Transition: The CO<sub>2</sub> Impact

The continuous growth in global energy demand, coupled with the worsening of environmental issues, has led to the establishment of international agreements aimed at limiting and reducing carbon dioxide (CO<sub>2</sub>) emissions into the atmosphere. The United Nations Paris Agreement set the ambitious goal of controlling the rise in the average global surface temperature of the Earth [1]. To achieve this target, global CO<sub>2</sub> emissions (GCE) must be reduced by 50–85% compared to the emission levels of the year 2000 [2,3] in order to counteract the opposing international trend shown in Figure 1 [4,5]. One of the promising

and implementable strategies is the set of technologies known as carbon capture, transport, and storage (CCTS). The aim is to capture carbon dioxide directly from the exhaust gases of industrial activities to prevent its release into the atmosphere, then compress and convert it into a fluid that can be transported and either reused in specific operations such as enhanced oil recovery (EOR) or stored in depleted oil or gas reservoirs. Therefore, the captured CO<sub>2</sub> must be safely transported to a storage facility or final utilization site in order to confine or use it without releasing it into the atmosphere. A key factor that could enable the rapid and large-scale implementation of CCTS is the repurposing of the existing underground pipeline infrastructure for CO<sub>2</sub> transport. Transporting CO<sub>2</sub> via pipelines has been recognized as the most efficient and economically viable method and has garnered significant attention from both industry and researchers. However, a key challenge identified in all analyses is ensuring adequate safety standards, not only during the capture and injection phases but particularly during transport across the territory. This involves studying the behaviour of the pipeline materials in contact with carbon dioxide in order to qualify them for such use. In particular, the composition of the transported mixture must be considered. Due to the nature of its source, the stream is not composed of pure carbon dioxide. The presence of other substances, such as non-condensable gases, residual unburned gases, and water vapour (which may condense), alters the mechanisms and severity of corrosion behaviour within the pipelines.



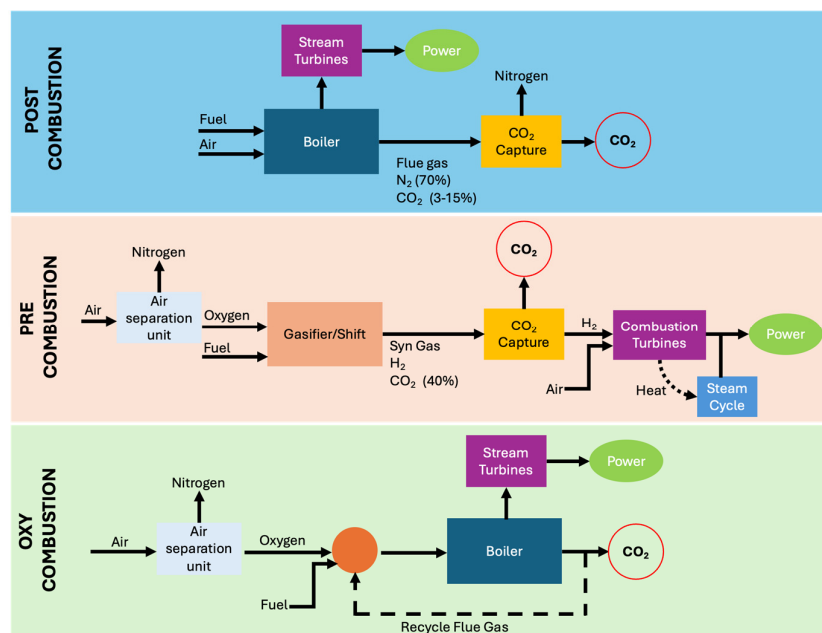
**Figure 1.** Global CO<sub>2</sub> emissions from fossil fuels by regions, 2000–2024, adapted from [5], showing emissions from the major countries and global regions.

The aim of this review was to analyse the current state of knowledge on carbon capture, transport, and storage (CCTS), with particular emphasis on the compatibility of existing pipeline network materials with CO<sub>2</sub> streams containing impurities, and how these may influence corrosion behaviour. The data considered in this review were collected through the systematic analysis of several databases, including Scopus, complemented by searches conducted using engines such as Google Scholar, with subsequent verification of the retrieved sources. The literature analysis was further supported by advanced tools for the graphical representation of recent research trends (e.g., Litmaps visualisations assisted by artificial intelligence).

### 1.2. CO<sub>2</sub> Capture Techniques

Currently, there are three main approaches for large-scale CO<sub>2</sub> capture, as illustrated in Figure 2: pre-combustion, post-combustion, and oxy-fuel combustion. Each of these

methods operates under different conditions, with distinct advantages and disadvantages, which are specifically outlined as follows.



**Figure 2.** Scheme comparing the principal CO<sub>2</sub> capture approaches, including post-combustion, pre-combustion, and oxy-fuel combustion technologies.

- **Pre-combustion:** This method involves removing CO<sub>2</sub> from fossil fuels upstream of the combustion process, resulting in a hydrogen-rich gas that can be used as a clean and versatile fuel in power generation plants or for alternative uses (transportation, basic chemicals, etc.). From an energy standpoint, the associated penalty is generally low, both because the process deals with limited flow rates and because it can operate under pressure, a condition that facilitates CO<sub>2</sub> capture and reduces the energy cost of regenerating the sorbent agents [6,7].
- **Post-combustion:** This approach involves removing CO<sub>2</sub> from flue gases after the combustion of the fuel source, e.g., extracting CO<sub>2</sub> from the exhaust gases of power plants and industrial facilities. Post-combustion systems are the most technologically mature, thanks to the experience gained in the oil and gas sector and small-scale gas treatment applications. They are best suited for retrofitting existing plants, provided there is sufficient space available, given the large volumes involved. The main disadvantages are the high capital costs associated with the need to treat large gas volumes as well as significant energy penalties due to the regeneration phase [6,7].
- **Oxy-fuel combustion:** This third approach, among the most promising today, involves combusting the fuel source with nearly pure oxygen instead of air. This results in a flue gas stream rich in CO<sub>2</sub> and free of nitrogen. After passing through pollutant removal units and a condensation section to remove water vapour, the stream can be sent to storage. This process already finds applications in the steel and glass industries and is currently being explored in power generation at the global level [6,7].

The first two strategies discussed are historically the most widely adopted, largely due to the fewer limitations associated with retrofitting existing plants, as they do not require changes to the oxidant. The oxidant (usually air) generally does not represent a cost in these systems, and from an economic sustainability perspective, it is difficult to justify its replacement with alternative fluids. The SWOT analysis (Strengths, Weaknesses, Opportunities, and Threats) [6], presented in Figures 3 and 4, compares pre- and post-

combustion capture, highlighting the respective advantages and disadvantages of these two technologies.

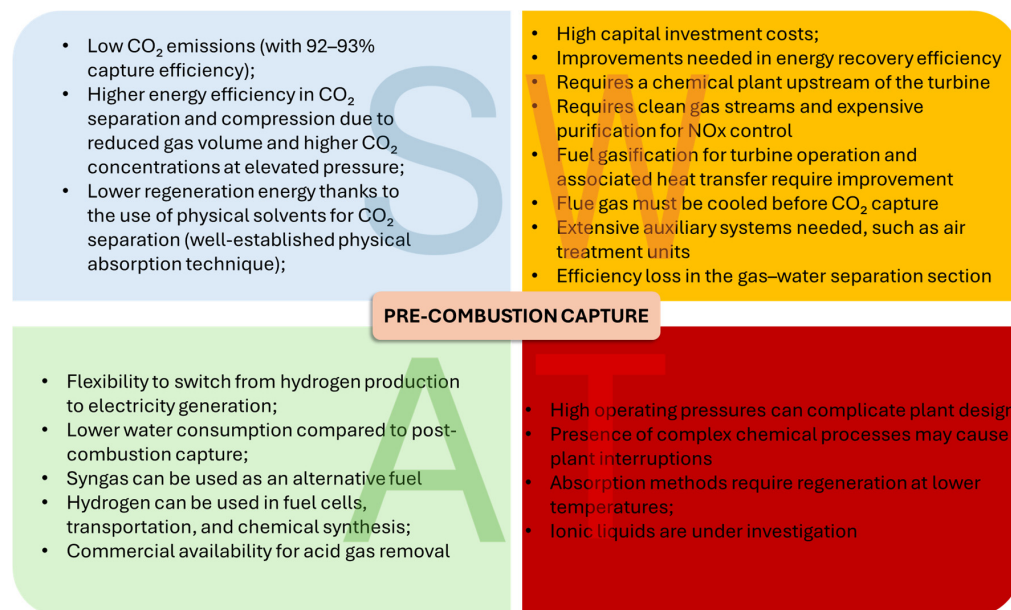


Figure 3. SWAT analysis for pre-combustion capture.

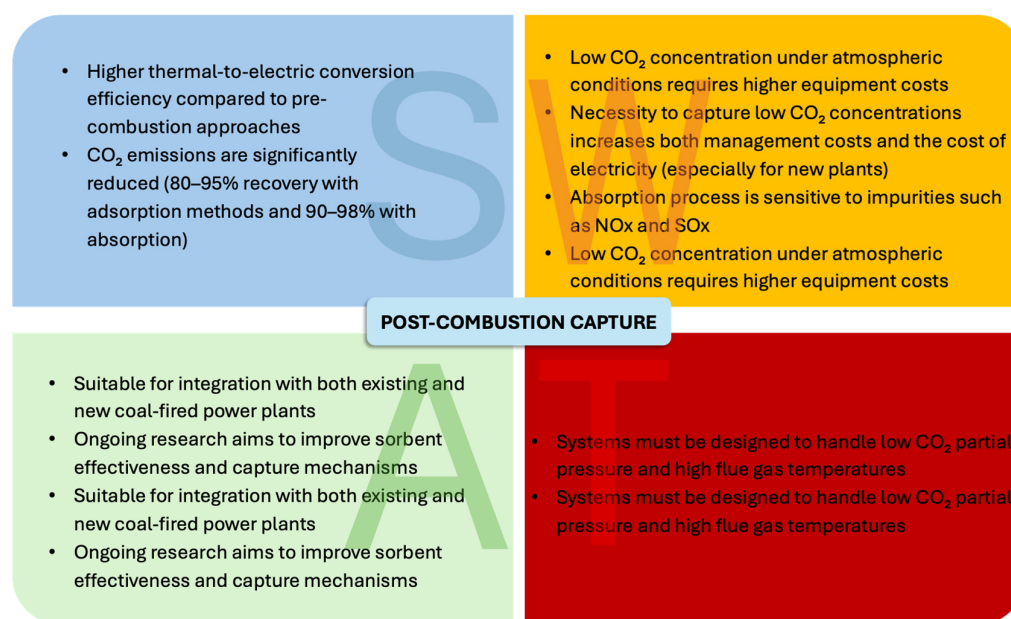


Figure 4. SWAT analysis for post-combustion capture.

## 2. Working Fluid Transported

### 2.1. Target Composition Criteria Within the Current Regulatory Framework

The widespread adoption of CCTS technologies has prompted standardization bodies to issue specific guidelines, such as those outlined in the ISO/FDIS 27913 [8]. These aim to regulate the operation and safety of such systems, with particular attention to plant management strategies, inspection scheduling, internal coating possibilities, and the composition of transported mixtures. All recommendations are intended to establish requirements that ensure safe operation of CO<sub>2</sub> transport systems. The ISO/FDIS 27913 framework seeks to establish guidance through generalised, integrable, and verifiable principles rather than highly prescriptive rules. It builds upon accumulated industry experience derived from

long-standing field applications, as reflected in earlier standards such as DNVGL-RP-F104 and the NORSOK standards, while simultaneously introducing broader and more theoretically grounded criteria. This ongoing evolution aims to strengthen consistency across practices and to promote harmonisation among standards within the oil and gas pipeline sector, thereby facilitating improved cooperation, interoperability, and integration across projects and regulatory frameworks. The standard highlights the well-known compatibility issues between the CO<sub>2</sub> and the steel typically used in natural gas pipelines. As a result, it proposes target compositions, presented in Tables 1 and 2, for gaseous and dense-phase CO<sub>2</sub>, respectively.

**Table 1.** Composition requirements for the CO<sub>2</sub> gas phase as specified by ISO/FDIS 27913, shown as a radial schematic representation of the allowable impurity distribution in the system.

Component	Hazard(s) in a CCU Context	Units	Limit
CO <sub>2</sub>	Asphyxiation, and can act as a toxicant at high concentrations	mol%	>95.0
N <sub>2</sub> <sup>a</sup>	Enhances the potential for ductile fracture Occupies store pore space inefficiently	mol%	≤4.0
H <sub>2</sub> <sup>a,b,c</sup>	Enhances the potential for ductile fracture and hydrogen induced crack propagation Affects the size of the multi-phase zone	mol%	≤1.0
Ar <sup>a</sup>	Occupies store pore space inefficiently, enhanced potential for running ductile fractures	mol%	≤4.0
CO <sup>a</sup>	Health and safety: toxic gas	mol%	≤0.2
Methane <sup>a</sup>	Occupies store pore space inefficiently	mol%	≤4.0
Ethane <sup>a</sup>	Occupies store pore space inefficiently	mol%	≤4.0
Propane and other aliphatic hydrocarbons <sup>d</sup>	Liquid drop-out is possible	mol%	≤0.15 in total
H <sub>2</sub> O	Enables corrosion of carbon steel	ppm mol	≤50
O <sub>2</sub> <sup>b,e</sup>	Enables oxidation of carbon steel Enhances bacterial growth in storage strata Other chemical reactions (e.g., with NO <sub>x</sub> , SO <sub>x</sub> , H <sub>2</sub> S)	ppm mol	≤10
NO <sub>x</sub> (NO, NO <sub>2</sub> ) <sup>f</sup>	Degradation of store caprock Takes place in the production of nitric and sulfuric acid	ppm mol	≤10
SO <sub>x</sub> (SO, SO <sub>2</sub> , SO <sub>3</sub> ) <sup>g</sup>	Degradation of store caprock Reactions with NO <sub>2</sub> can produce sulfuric acid	ppm mol	≤10
H <sub>2</sub> S <sup>h</sup>	Health and safety: toxic gas with foul odour	ppm mol	≤5
COS	Health and safety: toxic gas with foul odour	ppm mol	≤100
CS <sub>2</sub>	Health and safety: toxic gas with foul odour	ppm mol	≤20
NH <sub>3</sub>	Can react to form solid ammonium carbamate and other ammonium salts	ppm mol	≤10
BTEX <sup>i</sup>	Health and safety: toxic	ppm mol	≤15 in total
Methanol	Can introduce a liquid corrosive phase	ppm mol	≤350
Solid particulates <sup>j,k</sup>	Can reduce store permeability. Damage to compressor components	mg/Nm <sup>3</sup>	≤1 in total
Toxic metal <sup>j</sup>	Health and safety: toxic	mg/Nm <sup>3</sup>	≤0,15
VOCs <sup>l</sup>	Health and safety: toxic	mg/Nm <sup>3</sup>	≤48 in total

Table 1. Cont.

Component	Hazard(s) in a CCU Context	Units	Limit
Acid forming compounds <sup>m</sup>	Enables corrosion of carbon steel	mg/Nm <sup>3</sup>	≤150 in total
Amines <sup>n,o</sup>	Can introduce a liquid corrosive phase	ppb mol	≤100 in total
Glycols <sup>p</sup>	Enables aqueous corrosion of carbon steel		—
Nitrosamines and nitramines <sup>q</sup>	Health and safety: bio-toxic	µg/Nm <sup>3</sup>	≤3 in total
Naphthalene	Health and safety: toxic	ppb mol	≤100
Dioxins and furans <sup>r</sup>	Health and safety: toxic	ng/Nm <sup>3</sup>	≤0.02 in total

<sup>a</sup>→Combined total ≤ 5.0 mol%. <sup>b</sup>→Risk of acid drop-out with hydrogen > 100 ppm mol, if levels of SO<sub>2</sub>, H<sub>2</sub>S, O<sub>2</sub>, and NO<sub>2</sub> are much higher. <sup>c</sup>→Avoidance of SCC. <sup>d</sup>→Heavy hydrocarbons (C3+) shall not shift the dew point below that of pure CO<sub>2</sub>. <sup>e</sup>→Presence of O<sub>2</sub> influences the formation of strong acids and elemental sulphur, and increases the sensitivity to sulphur-induced stress corrosion cracking. <sup>f</sup>→Separating out the different components of NO<sub>x</sub> can allow higher levels of some species. <sup>g</sup>→Separating out the different components of SO<sub>x</sub> can allow higher levels of some species. <sup>h</sup>→H<sub>2</sub>S tends to form SO<sub>2</sub> and can form elemental sulphur, reacting with O<sub>2</sub> if present in sufficient levels. <sup>i</sup>→Separating out the different components of BTEX can allow higher levels of some species. <sup>j</sup>→The maximum size of the particulate is 1 µm. <sup>k</sup>→To include: Ash, dust, Na, K, Mg, Cr, Ni, Cd, Hg, Tl, Pb, As, and Se. <sup>l</sup>→To include: formaldehyde, acetaldehyde, dimethyl sulphide, ethanol. <sup>m</sup>→To include: Cl<sub>2</sub>, HF, HCl, HCN. <sup>n</sup>→The maximum size of the liquid droplet is 2 µm. <sup>o</sup>→To include: MEA, MDEA, DEA, AMP, piperazine, and any proprietary mixture containing any amine. <sup>p</sup>→To include: TEG, MEG, DEG, propylene glycol, dimethyl ethers of polyethylene glycol. <sup>q</sup>→To include: NDMA, NMEA, NDEA, NDELA, NPIP, NMor. <sup>r</sup>→To include: PCDD, PCDF.

The focus is on components known to exacerbate corrosion mechanisms, such as the presence of H<sub>2</sub>S, SO<sub>x</sub>, NO<sub>x</sub>, non-condensable residues like CH<sub>4</sub> and C<sub>2</sub>H<sub>6</sub>, and water. Beyond the compositional limits listed in the tables, the standard specifies the following: “The CO<sub>2</sub> stream shall not contain impurities that may cause harm or damage to the pipeline, equipment, downstream systems, or storage reservoirs. Impurities not listed shall be brought to the attention of the pipeline operator for further assessment. CO<sub>2</sub> streams from anthropogenic sources that do not meet specifications may be rejected at the discretion of the pipeline operator.”

Given the nature of the process and the fluid involved, it is therefore possible to encounter conditions in which the composition may be highly variable, depending on the type of upstream plant, the technology used for carbon dioxide capture, and the specific transport facility. In general, the fluids to be treated and transported may contain the following:

- Large volumes of O<sub>2</sub>, N<sub>2</sub>, H<sub>2</sub>, Ar, and H<sub>2</sub>O (in the form of vapour or condensate);
  - Particulates/dust that need to be mechanically separated (and which are generally absent during transport);
  - Traces of components that may act as catalyst poisons or be toxic to downstream process microorganisms, or even for human consumption. For example, acidic components such as H<sub>2</sub>S, SO<sub>2</sub>, SO<sub>3</sub>, HCl, HF, COS, CS<sub>2</sub>, CH<sub>3</sub>SH, HCN, NO, NO<sub>2</sub>/NO<sub>3</sub>, and Cl<sub>2</sub>; basic components such as NH<sub>3</sub> and amines, whose presence is due to the flue gas purification process (amine-based processes being among the most common); combustible components such as CO, CH<sub>4</sub>, and organic substances; metallic components such as mercury, heavy metals (Ni, Cr, etc.), and alkali and alkaline earth metals (Na, K, Ca, Ba), which occur in the form of aerosols rather than particulates; volatile organic compounds such as aromatic hydrocarbons, olefins, aldehydes/organic acids, dioxins/furans, oils/greases, etc.

**Table 2.** Composition requirements for CO<sub>2</sub> in the dense phase as specified by ISO/FDIS 27913 including the allowable limits for impurities.

Component	Notes	Units	Limit
CO <sub>2</sub>	Dry basis	mol%	>95.0
N <sub>2</sub>		mol%	a,b
H <sub>2</sub>		mol%	≤1
Ar		mol%	a,b
CO	Total non-condensables to be <5 mol%	mol%	≤0.7
Methane		mol%	
Ethane		mol%	
Propane and other aliphatic hydrocarbons	Total hydrocarbons to be <5 mol% and a dew point of product with respect to hydrocarbons to be <−20 °C.	mol%	≤1
H <sub>2</sub> O	The limit for water may be higher (e.g., 630 ppm mol) if the CO <sub>2</sub> stream contains very low levels of O <sub>2</sub> , NO <sub>x</sub> , and SO <sub>x</sub> (e.g., geological CO <sub>2</sub> ). <sup>b</sup>	ppm mol	≤100
O <sub>2</sub>		ppm mol	≤10
NO <sub>x</sub> (NO, NO <sub>2</sub> )		ppm mol	≤1.5
SO <sub>x</sub> (SO, SO <sub>2</sub> , SO <sub>3</sub> )		ppm mol	≤1
H <sub>2</sub> S		ppm mol	≤55
Total sulphur		ppm mol	≤50
Solid particulates		ppm wt	≤1
Mercury		ng/L	≤5
Amines		ppm wt	≤1
Glycols	Must not be present in a liquid state at the temperature and pressure conditions of the pipeline.	ppm mol	≤50
Compressor lube oil carry-over		ppm wt	≤50 ppmw
Liquids	CO <sub>2</sub> stream shall be free of liquids at delivery conditions and shall not produce condensed liquids in the pipeline at pipeline temperature and pressure.		

a→Impurities causing harm of damage to pipelines, equipment, downstream systems, or reservoirs. b→It is possible, with a water content of 100 ppm mol, for water drop-out to take place during depressurization (e.g., for maintenance). If this operation is planned, then a gas phase specification should be considered to avoid aqueous phase formation.

Moreover, the difficulty in maintaining unchanged composition over time and along the transport route as well as the variable temperature and pressure conditions influence metal–environment interfacial phenomena; the physical properties of pure carbon dioxide are altered by the presence of the aforementioned contaminants, resulting in a variation of the fluid’s properties, which may cause phase transitions at temperatures and pressures different from those of pure CO<sub>2</sub>, thus altering the characteristic corrosion mechanisms (generally worsening their severity). It is therefore appropriate to describe the transport fluid as a CO<sub>2</sub> mixture.

The phase transition phenomena of the fluid are thus complex and difficult to predict. In particular, the standard suggests that “transient situations such as pipeline shutdown and cooling (especially in transport systems where a CO<sub>2</sub> flow enters the pipeline at temperatures higher than ambient) must be considered. [Phase transition phenomena] may

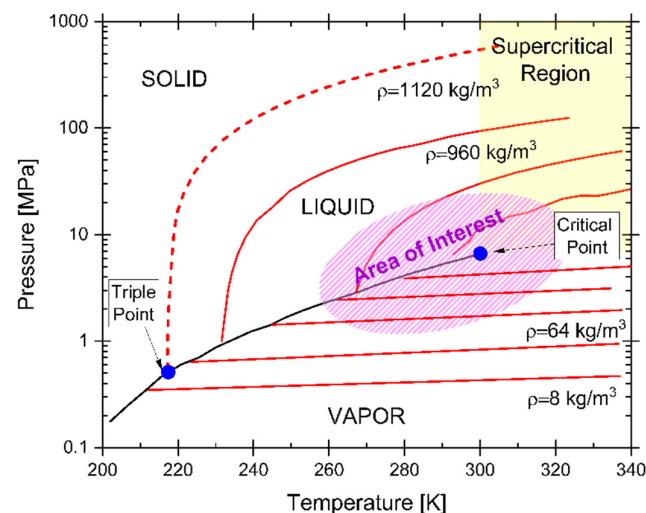
also occur during depressurization of a dense-phase transport system. Following such events, multiphase flow conditions are inevitable, and the operator must strive to minimize their duration.”

As described in the standard, the main aspects to be considered in the management of a pipeline under multiphase flow conditions are as follows:

- The maintenance of stable conditions; this may involve adjusting the injection capacity of the system (pipeline and well geometry, reservoir conditions, ambient temperature, compression and reservoir conditions, ambient temperature, compression and pumping equipment, etc.)
- The identification of flow conditions that can reduce hydraulic capacity (hydrate formation) and compromise system integrity (erosion, hydrate formation, corrosion potential, etc.)
- Maintaining temperature within an acceptable range; pressure reductions resulting from normal operations may lead to temperature drops due to evaporation of liquid CO<sub>2</sub>, affecting heat transfer and pipelines.

## 2.2. Phase Transitions and CO<sub>2</sub> Phase Diagram

Even under controlled conditions, due to natural variations in temperature and pressure along the transport line and the variability of the incoming mixture, the fluid may undergo phase changes during pipeline transport. In particular, carbon dioxide can experience phase transitions along the pipeline, resulting in a biphasic state, e.g., a coexistence of liquid and gaseous phases. The morphology of the phase diagram is therefore an essential tool for predicting the behaviour of the transported fluid (and consequently its corrosion behaviour). The phase behaviour of pure carbon dioxide has been extensively studied and serves as a reference point. Figure 5 shows an example, including the vapour–liquid equilibrium curve, which extends from the triple point (216.59 K and 0.518 MPa) to the critical point (304.13 K and 7.377 MPa). A highly accurate equation of state for CO<sub>2</sub> was proposed by Span and Wagner [9], valid over a wide range from 217 K to 1100 K and up to 800 MPa. Subsequently, several thermodynamic data sources have been published to further characterize CO<sub>2</sub> behaviour under different conditions [10–13].

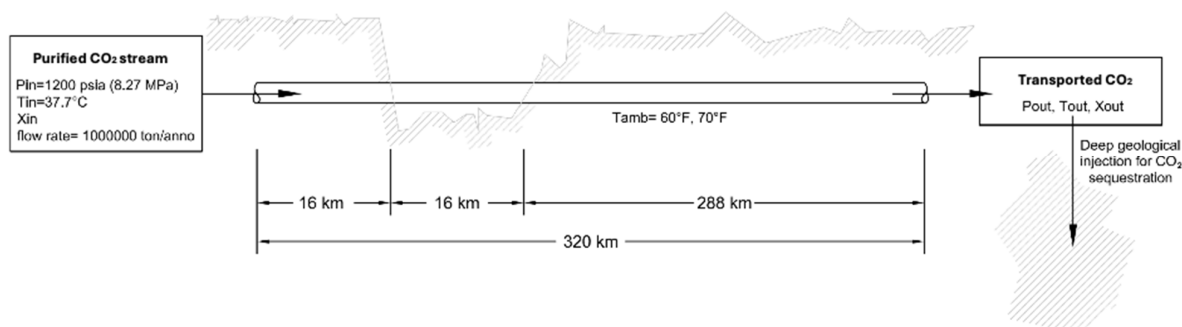


**Figure 5.** Phase diagram of pure CO<sub>2</sub>, illustrating phase boundaries between solid, liquid, and gaseous states as a function of temperature and pressure.

In the case study examined in this review, the pipelines considered are onshore buried pipelines, for which an average subsurface temperature of approximately 15–18 °C can be assumed, with internal pressures of around 40 bar. For offshore sections, the operating

pressures are significantly higher (ranging from 100 to 300 bar), while the temperature remains relatively constant.

The literature provides several examples of operating parameters plotted on the phase diagram of pure CO<sub>2</sub>, which allow for optimal transport conditions [14], as shown in Figure 6. The graph presents the pressure and temperature conditions across the entire CO<sub>2</sub> transport, storage, and utilization chain [3,13,15–17], operating in the so-called supercritical state (s-CO<sub>2</sub>). This specific condition ensures that the amount of CO<sub>2</sub> transported per unit volume is maximized, as the fluid exhibits liquid-like density and gas-like viscosity [18]. For instance, the density and viscosity of s-CO<sub>2</sub> at 50 °C and 10 MPa are approximately 395 kg/m<sup>3</sup> and 0.03 cP, respectively [19].



**Figure 6.** Schematic representation of the pipeline system analysis used in the model.

For cost-effective CO<sub>2</sub> transport, pipelines should ideally operate within a range of 20–50 °C and 5–15 MPa. However, ensuring full compatibility between these transport conditions and existing infrastructure can be challenging. In fact, the operating pressure range of conventional oil and gas pipelines is significantly lower than the optimal range for CO<sub>2</sub> transport as reported in literature [17,20–22]. Nevertheless, altering the fluid’s operating conditions would reduce the economic viability of the transport process. Captured CO<sub>2</sub> from combustion facilities is typically compressed above its critical point (7.4 MPa, 31 °C) to avoid biphasic flow regimes and thereby achieve optimal transport efficiency.

### 2.3. Effects of Impurities on the Physical Properties of the Fluid and on Facilities

The separation of impurities, if it is possible, represents both a cost and a complication in the CCTS process. For this reason, it would be reasonable to consider the direct injection of CO<sub>2</sub> containing impurities into underground storage sites. Moreover, some emerging technologies aim at the co-capture and co-storage of multiple atmospheric pollutants alongside CO<sub>2</sub>, thereby reducing the extent of required treatment. However, the presence of impurities in the transported CO<sub>2</sub> stream can significantly impact both pipeline network performance and corrosion behaviour within the infrastructure.

Impurities alter the physical properties of the fluid, consequently modifying the phase diagram and the critical transition values compared to those of pure CO<sub>2</sub>, as previously illustrated in Figure 6. Non-condensable impurities such as N<sub>2</sub>, O<sub>2</sub>, and Ar raise the saturation pressure of liquid CO<sub>2</sub> and lower its critical temperature. As a result, a lower temperature and an additional overpressure are required to avoid undesirable biphasic flow during CO<sub>2</sub> transport in pipelines. The type of pollutants present in the CO<sub>2</sub> stream can therefore shift the operating pressure range, increasing the risk of entering biphasic regimes, as demonstrated by the phase envelopes of CO<sub>2</sub> mixtures [23].

Indeed, studies in the literature have reported greater pressure drops during the pipeline transport of CO<sub>2</sub> mixtures (e.g., impure CO<sub>2</sub>) compared to pure CO<sub>2</sub> [18,24–27]. Naturally, the impurity levels depend on the capture and separation technology employed, as well as the CO<sub>2</sub> source. CO<sub>2</sub> derived from an Integrated Gasification Combined Cycle

(IGCC) plant followed by a conventional amine scrubbing system tends to be contaminated mainly with carbon monoxide and hydrogen sulphide. In contrast, CO<sub>2</sub> from oxy-fuel combustion processes is likely to exhibit elevated levels of argon, oxygen, and potentially nitrogen.

### 2.3.1. Variation of the Critical Point: Canadian Case Study on Impurities

A relevant study on the transport of CO<sub>2</sub> mixtures applied to real-case scenarios was conducted by Natural Resources Canada (NRCan), which has led several federal CCS programs. In addition to research and development projects focused on CO<sub>2</sub> capture from coal-fired power plants and other industrial sources, NRCan has been involved in CO<sub>2</sub> storage research initiatives, including injection, monitoring, measurement and verification, storage integrity assessment, and CO<sub>2</sub> capacity estimation.

In this context, NRCan carried out a study supported by the International Energy Agency (IEA) and the U.S. Greenhouse Gas Center (US GHG), focusing on the effects of impurities in CO<sub>2</sub> streams, with particular emphasis on saline formation storage technology, a strategy that appears to hold considerable promise [23].

The study analysed five different impurity mixtures combined with CO<sub>2</sub> from various industrial sources and compared their behaviour to that of pure CO<sub>2</sub>, as summarized in Table 3.

**Table 3.** List of CO<sub>2</sub> mixtures studied by Natural Resources Canada (NRCan) from different plants.

No.	Source and Type of Plant	Transported Mixture	Reference
1	CO <sub>2</sub> stream from oxy-fuel combustion in a fluidized bed combustor at the CanmetENERGY pilot plant	5.2 vol% O <sub>2</sub> , 221 ppm CO, 1431 ppm SO <sub>2</sub> , 243 ppm NO	[28]
2	CO <sub>2</sub> stream from a zero-emission process proposed by CanmetENERGY	1.05% CO, 1.7% SO <sub>2</sub> , 0.32% H <sub>2</sub> , 690 ppm H <sub>2</sub> S	[29]
3	CO <sub>2</sub> stream from the Cansolv <sup>®</sup> absorption system	2.9% SO <sub>2</sub>	[30], as reported in a previous IEA GHG report
4	Predicted CO <sub>2</sub> stream from a pre-combustion capture plant	1 vol% H <sub>2</sub> , 0.9 vol% N <sub>2</sub> , 300 ppm Ar, 100 ppm H <sub>2</sub> S, COS and other impurities	Composition data provided by IEA GHG [30]
5	Predicted CO <sub>2</sub> stream from an oxy-fuel combustion plant	5.8 vol% N <sub>2</sub> , 4.7 vol% O <sub>2</sub> , 4.47 vol% Ar, 100 ppm NO <sub>x</sub> , 50 ppm SO <sub>2</sub> , 20 ppm SO <sub>2</sub> , 50 ppm CO	Composition data provided by IEA GHG [30]

In this study the behaviour of the various fluids was analysed using the Peng–Robinson equation of state, which is now widely adopted for estimating the phase envelope shifts of CO<sub>2</sub> mixtures [31]. The results show that the critical temperature and pressure of the mixtures were different from those of pure CO<sub>2</sub>. The N<sub>2</sub>, O<sub>2</sub>, Ar, and H<sub>2</sub> exhibit a pronounced effect, leading to an increase in the saturation pressure of the liquid phase and a reduction in the critical temperature. An extreme case is represented by mixture 5

(from Table 3), derived from combustion sources. In this scenario, the critical temperature decreases by approximately 10 °C, and the liquefaction pressure increases by more than 50 bar, compared to pure CO<sub>2</sub>. On the other hand, SO<sub>2</sub> has the opposite effect: it leads to a reduction in saturation pressure and an increase in critical temperature, consistent with the high critical temperature of pure SO<sub>2</sub> (157.6 °C). It is also worth noting that impurities present at low concentrations, such as CO and NO<sub>x</sub>, do not significantly influence the phase behaviour of CO<sub>2</sub>.

### 2.3.2. Phase Diagram Variation: The Role of Water

Among all potential impurities, free water (H<sub>2</sub>O) is the most undesirable due to its capacity to induce hydrate formation and corrosion issues, particularly when associated with acidic gaseous components such as CO<sub>x</sub>, SO<sub>x</sub>, and H<sub>2</sub>S. When the water content exceeds its maximum solubility, it can form hydrates that potentially lead to pipeline blockages [3]. Moreover, even when water is the only impurity present, condensation due to temperature and pressure fluctuations along the pipeline can cause severe corrosion phenomena. It is well known that water presence is critical for corrosion processes in CCTS and EOR transport components, even under supercritical CO<sub>2</sub> (s-CO<sub>2</sub>) conditions [32]. Consequently, it is essential to purify, dehydrate, and compress the gaseous CO<sub>2</sub> prior to pipeline transport, as recommended by ISO/FDIS 27913, and to strictly control water concentrations in the transported fluid. Water is commonly present in all purified CO<sub>2</sub> streams. After compression, the water content equilibrates to a saturation level determined by temperature and pressure. For example, if a CO<sub>2</sub> stream is compressed from ambient pressure to 200 bar and then cooled to 35 °C for deep injection, the resulting mole fraction of water is approximately 0.0015 (1502 ppm) [25]. The water content can be reduced to acceptable levels through appropriate dehydration techniques prior to pipeline transport. The system composed by CO<sub>2</sub> and water exhibits a broad region of fluid immiscibility and solid–fluid equilibria. The phase equilibrium conditions as functions of pressure (P), temperature (T), volume (V), and absolute humidity (X), defined as the mass of water vapour per volume of mixture, have been extensively studied and reported in numerous publications [33–39]. Recent thermodynamic property measurements for this system include the works of Novitskiy et al. [40] and Siqueira-Campos et al. [41]. The GERG-2008 equation of state (EoS) is the most widely used model for predicting the physical behaviour of the biphasic CO<sub>2</sub>–water system [42]. However, it is based solely on density data, which limits its accuracy in estimating phase equilibria particularly at the lower boundary to low relative humidity value. Furthermore, it has been estimated that if water concentrations at the compressor outlet exceed  $X_{CO_2} \approx 0.001$ , CO<sub>2</sub> hydrate formation can occur in some sections of the surface infrastructure [33,38]. These hydrates may accumulate and, upon melting, locally increase the water concentration to levels where liquid water can collect in crevices, posing a significant corrosion hazard.

### 2.3.3. Variation of Physical Properties of CO<sub>2</sub> Mixtures Induced by Pipeline Transport: Pipeline Model

The undesirable phase transitions in a CO<sub>2</sub> and pollutant mixture can be induced by the transport of the fluid itself along the pipeline. In fact, the inlet values of pressure, temperature, and relative humidity can vary due to pressure drops and the different positioning of the pipelines along the transport path. Therefore, the thermodynamic parameters that characterize the state of the fluid may change, causing state variations that can trigger phase transitions, which are often difficult to predict.

An interesting case study was analysed by Verma et al. [25], simulating the behaviour of a system with a purified carbon dioxide stream which, after compression and cooling in

a post-cooler, must be transported and injected deep underground for CO<sub>2</sub> abatement. The summarised scheme of the system simulated is shown in Figure 6.

The inlet values at the pipeline are considered known: pressure ( $P_{in} = 82.7$  bar), temperature ( $T_{in} = 37.7$  °C), vapour fraction, and mass flow rate ( $m = 1,000,000$  tons/year). The room temperature is  $T_{amb} = 15.6$  °C, and the case for  $T_{amb} = 21$  °C is also analysed. Pressure drops due to friction in the pipeline are calculated using the Beggs and Brill correlations as well as the OLGA S. model.

For pure CO<sub>2</sub>, under supercritical inlet conditions and moderate ambient temperatures (15.6 °C), the fluid transitions to the liquid phase with relatively minor pressure drops and no vapour formation. The temperature of the CO<sub>2</sub> stream decreases progressively along the pipeline until reaching ambient conditions. Under these ambient conditions, a pressure reduction is observed along the pipeline relative to the inlet. Phase change occurs only when the ambient temperature is increased, with the vapour fraction exceeding 40% at the outlet [25]. In contrast, for the CO<sub>2</sub> + 4 mol% H<sub>2</sub>S mixture at a reduced inlet pressure (7.58 MPa), the phase behaviour alters significantly even at lower pressures compared to the pure CO<sub>2</sub> case. Although the mixture initially enters the pipeline in a supercritical state, it transitions into the vapour–liquid region along the pipeline length. The vapour fraction increases progressively due to the combined effects of pressure reduction and the latent heat of vaporisation. This results in further cooling and increases frictional pressure losses, which in turn promote additional vapour formation through a positive feedback mechanism. Such behaviour is not observed in the pure CO<sub>2</sub> case at comparable pressures, underscoring the critical influence of impurities on two-phase flow development and pipeline performance. Consequently, when the CO<sub>2</sub> stream intersects the vapour–liquid boundary or enters the vapour region, frictional pressure losses become substantial and potentially prohibitive. The addition of small amounts of impurities such as H<sub>2</sub>S, N<sub>2</sub>, Ar, and O<sub>2</sub> to CO<sub>2</sub> does not significantly expand the pressure–temperature (P–T) immiscibility region [25]. Consequently, H<sub>2</sub>S does not cause notable variations in phase equilibrium effects, critical pressure changes, or compression requirements along the pipeline section.

This model was also applied to a CO<sub>2</sub>–water mixture. However, since the water content in the CO<sub>2</sub> stream was limited to 0.0032 mol%, no impact of water vapour on the temperature or pressure profiles along the pipeline was observed. Therefore, based on the reviewed literature and under the proposed conditions, this value can be considered a threshold below which the thermodynamic parameters and the predicted fluid behaviour in the pipeline are comparable to those of pure CO<sub>2</sub>. Generally, the concentrations of impurities present in a CO<sub>2</sub> mixture can vary significantly, primarily due to the diversity of the sources from which the streams to be transported originate. Compositional variation, especially under near-critical conditions, strongly influences pressure drop. Therefore, the models employed must consider the composition and the corresponding equation of state for operational design. The thermodynamic properties of pure CO<sub>2</sub> are accurately described by the GERG-2008 equation of state (EoS). However, the behaviour of CO<sub>2</sub> + H<sub>2</sub>S mixtures (and impurities in general) is difficult to estimate because data for this system at pressures above 20 MPa are scarce or unavailable. Consequently, GERG-2008 currently does not accurately reproduce fluid equilibria at low temperatures below 373 K, although sufficient data exist to potentially re-parameterize the model.

### 3. Phenomena of Corrosion Associated with CO<sub>2</sub> Transport

CO<sub>2</sub> streams transported at low to medium pressure in metallic pipelines can trigger corrosion phenomena known as sweet corrosion. These mechanisms have been studied for over 50 years, and there are now highly complex predictive models to estimate corrosion rates based on CO<sub>2</sub> partial pressure, temperature [43–45], pH, dissolved salts, and the

formation of more or less protective scales [46–48]. However, recent studies [49] have emphasized that these models are ineffective in predicting corrosion rates in the presence of high CO<sub>2</sub> pressures and contaminants. Additionally, low-pressure and low-temperature operational conditions, as well as compositional changes associated with thermodynamic parameter variations along the transport, can cause condensates to accumulate and trigger sweet corrosion phenomena known as “top of the line corrosion” (TLC). Condensates of water vapour containing CO<sub>2</sub> and other contaminants may form on the cooled inner walls of pipelines, thereby initiating aggressive corrosive processes [49–51].

### 3.1. The Role of Water in the Corrosion of Transported CO<sub>2</sub> Mixtures

A key aspect in analysing sweet corrosion and TLC phenomena is the presence of water within transported fluids. The CO<sub>2</sub> corrosion process generally follows three phases:

1. Dissolution of CO<sub>2</sub>
2. Hydration of CO<sub>2</sub>
3. Dissociation of carbonic acid

#### 3.1.1. Dry CO<sub>2</sub> Flow (Absence of Water)

A primary distinguishing factor is whether water is present (in vapour–liquid or condensate biphasic conditions) within the transported CO<sub>2</sub> flow, which alters the corrosion mechanisms. The influence of protective corrosion-product formation in an aqueous CO<sub>2</sub> environment is highly complex, although the underlying chemical reactions in an H<sub>2</sub>O/CO<sub>2</sub> system remain the same regardless of CO<sub>2</sub> partial pressure [52,53].

In the literature, it has been widely demonstrated that dry CO<sub>2</sub>, that means with total absence of water, is not corrosive in an engineering sense. Numerous tests estimate corrosion rates under such conditions, for example the following:

- Short-term exposure tests [54,55]
- 24 h at 50 °C and 24 MPa on C1018 carbon steel [56,57]
- 96 h in supercritical dry CO<sub>2</sub> at 80 °C and 12.5 MPa [57]
- 200 days at 9–12 MPa and 160–180 °C on AISI 1080 carbon steel yielded a corrosion rate of ~0.01 mm/year [58]
- After 12 years of service in high-pressure dry CO<sub>2</sub> pipelines, carbon steel corrosion rates were 0.5–2.5 µm/year [59,60].

Flow-imposed experiments (e.g., 100 rpm at ~8 MPa and 35 °C) showed that corrosion rates for stainless steels 304 L and 316 L were 0.0008 and 0.0012 mm/year, while carbon steels X42 and X60 reached ~0.02 mm/year [61]. Therefore, carbon steels in a predominantly dry CO<sub>2</sub> environment exhibit minimal corrosion. This conclusion is further supported by conductivity measurements (~3 × 10<sup>-5</sup> S/m) of two orders of magnitude lower than those in water-saturated supercritical CO<sub>2</sub> (~7 × 10<sup>-3</sup> S/m) [62], indicating negligible corrosion in pure dry CO<sub>2</sub>. In summary, carbon steels exhibit negligible or no corrosion when exposed to dry CO<sub>2</sub> environments.

#### 3.1.2. CO<sub>2</sub> Flow in the Presence of Water

##### Mixtures of CO<sub>2</sub> with Small Quantities of Water (Below Solubility Limit)

The behaviour of pipelines transporting CO<sub>2</sub> with water present was analysed through USA field studies conducted from 1990 to 2001. These studies reported only two incidents attributable to corrosion phenomena [16], with generally low corrosion rates of ~0.00025–0.0025 mm/year [54,56]. Strict limits were imposed on allowable contaminants, particularly free water concentrations. For example, steel pipeline X60 exhibited corrosion rates < 0.001 mm/year at 14 MPa with 800–1000 ppm H<sub>2</sub>O [52]. In another study, a mixture

of 600 g supercritical CO<sub>2</sub> with 1.5 g H<sub>2</sub>O at 80 °C for 96 h yielded a corrosion rate of 0.0053 mm/year on X65 carbon steel [63].

Thus, if the water content remains below its solubility limit in CO<sub>2</sub>, no separate or water-rich aqueous phase forms. As a result, CO<sub>2</sub> does not significantly dissolve in water, and carbonic acid (H<sub>2</sub>CO<sub>3</sub>), the primary driver of corrosion in ferrous materials, is not generated. In this condition the carbon steels do not experience significant corrosion under relevant pressure and temperature conditions [64–68].

Conversely, exceeding the solubility limit leads to rapid increases in corrosion rates due to liquid water phase presence in the CO<sub>2</sub>/H<sub>2</sub>O system. For example, mixtures with substantial water content (600 g supercritical CO<sub>2</sub> and 100 g H<sub>2</sub>O [63]) at 50–130 °C for 96 h resulted in corrosion rates of 0.014–0.043 mm/year, nearly ten times higher than lower-water mixtures. Insufficient drying of liquid CO<sub>2</sub> can lead to residual H<sub>2</sub>O accumulation, causing carbon steel corrosion and pipeline leaks [64]. Even with 100 ppm water, severe corrosion rates (up to 1.2 mm/year) were observed where residual water pools on carbon steel surfaces, creating aggressive local environments [64].

The corrosion rate data from various researchers [56,61,63,65,67,69–76] are summarised in Table 4, illustrating the extent of corrosion in steels across different water contents in supercritical or aqueous CO<sub>2</sub> environments. The key conclusions drawn are as follows:

- Carbon steels and Cr-containing stainless steels show negligible or undetectable corrosion rates (<0.001 mm/year) in the absence of water
- Carbon steels incur mild corrosion (<0.1 mm/year) when water content in supercritical CO<sub>2</sub> remains below saturation (undersaturated conditions)
- Carbon steels suffer corrosion rates exceeding 0.1 mm/year, (and even reaching 1 mm/year) when in water-saturated supercritical CO<sub>2</sub>.

**Table 4.** Summary of CO<sub>2</sub> corrosion rates with different water concentrations from the literature (adapted by [32]).

N°	P (bar)	T (°C)	H <sub>2</sub> O (ppmv)	Material	t (h)	Flow	Corrosion Rate (mm/year)	Refs.
1	80	40	244	C-steel	168	Static	0.08	[72]
11	80	50	Sat	X65	14, 24, 48	Static	0.024 ≈ 0.1	[71]
26	79.6–82	35	10 g	SS: 304 L, 3161 C-Steel: X42, X60	120	100 rpm	SS: 0.0005–0.0008 C Steel: 0.007	[61]
29	80	50	650 (Undersat)	X65	24	Static	No corrosion	[64]
30	80	50	3310 (Sat)	X65	24	Static	0.38	[64,65,76]
31	80	50	10 g (Sat.)	X65	24	Static	0.4–1	[65]
32	75.8	40	244	C Steel	5	Static	1.2	[73,74,77]
35	79	31	244	C steel	5	Static	1.1	[73]
37	95 182	50–130	100 g (Sat.)	C steel	96	995	0.014–0.043	[63]
39	100	20	1220	X65	720	Static	No corrosion	[78]
45	100	25	488 e 1222	X65	336	3 rpm	0	[67]
46	125	80	1.5 g	38Mn6/C75	96	995	0.0036	[63]
49	123–146	25–60	Saturated	X65	48–400	180 rpm	0.01–0.1	[75]
50	240	50	40 g (Sat.)		24	Static	Not given	[56]

#### Mixtures of CO<sub>2</sub> with Saturated Aqueous Phase

As the water content in the CO<sub>2</sub>/H<sub>2</sub>O mixture increases, the system gradually transitions from a CO<sub>2</sub>-rich phase to a H<sub>2</sub>O-rich phase. This process can be supported by the mutual dissolution of supercritical CO<sub>2</sub> (s-CO<sub>2</sub>) into H<sub>2</sub>O and vice versa. When the water content in CO<sub>2</sub> is high, the primary phase of the system consists of free water. Supercritical

CO<sub>2</sub> conditions have been widely studied in the literature, and their behaviour can be extrapolated to cases of CO<sub>2</sub> transport under non-supercritical conditions.

To compare the difference in corrosion rates of carbon steels in s-CO<sub>2</sub> environments between CO<sub>2</sub>-rich and H<sub>2</sub>O-rich phases, the corrosion behaviour of carbon steel exposed for 24 h at 50 °C under pressures ranging from 4.6 to 8 MPa was analysed in environments with either predominant s-CO<sub>2</sub> or predominant H<sub>2</sub>O content [76]. It was found that in both environments, the corrosion rate does not significantly depend on CO<sub>2</sub> pressure, but rather on the presence of water. Specifically, in a CO<sub>2</sub>-rich environment, the corrosion rate is approximately 0.4 mm/year, whereas in an H<sub>2</sub>O-rich environment, the rate reaches about 20 mm/year. The corrosion rates of carbon steels under these conditions have been reported by various researchers, as reported in Table 5 [63–65,75,76,79,80]. It can be observed that the corrosion rate in H<sub>2</sub>O-rich systems was nearly two orders of magnitude higher than in CO<sub>2</sub>-rich systems.

**Table 5.** Comparison of corrosion rates from literature under s-CO<sub>2</sub> and water-saturated conditions.

Environment	Corrosion Rate (mm/years)	Refs.
H <sub>2</sub> O-rich phases	19.2	[63–65,75]
	10.6	[75]
	5–15	[32]
	0.6	[78]
	1.7	[78]
	5–30	[81,82]
CO <sub>2</sub> -rich phases	0.38	[75]
	0.013–0.043	[32]
	0.04	[78]
	0.01–0.1	[81,82]

However, literature sources [70,81] have also shown that the corrosion rate increased when the water content was raised from 500 to 1000 ppmv in flow loop tests. Interestingly, it was also reported that further increasing the water content from 1000 to 2000 ppmv resulted in a decrease in the corrosion rate.

Under s-CO<sub>2</sub> conditions, the N.O.R.S.O.K. [79] and K.S.C. [83] models (commonly used to predict carbon steel corrosion rates in CO<sub>2</sub> environments) are no longer applicable. For example, in studies involving X65 steels [80], the N.O.R.S.O.K. and K.S.C. models estimated corrosion rates of 10 to 17 mm/year, whereas experimental tests reported rates between 1 and 6 mm/year. The values predicted by the two models are significantly higher than those measured and are considered overly conservative.

### 3.2. Corrosion Mechanism: Top of the Line Corrosion (TLC)

Top of the line corrosion (TLC) operates through a complex mechanism that results in severe localized corrosion, frequently affecting critical components. In CO<sub>2</sub>-dominated environments, the types of localized attack are commonly the following:

- pitting corrosion, when attack initiates at small defects in the passive corrosion-product layer where the base metal interfaces with the environment
- mesa corrosion, when steel is not corroded uniformly but presents surface pits often wide and flat-bottomed, surrounded by corrosion products [84].

Field applications show that carbon-steel pipelines used for CO<sub>2</sub> transport typically have inner diameters ranging from 14 to 22 inches. The lower portion (approximately one-third of the external surface) is buried in soil or seabed and is often thermally insulated, whereas the upper part may not be buried and thus inadequately insulated.

A typically hotter fluid flows inside the pipeline, while along the uninsulated top, heat exchange results in condensate formation. Internal temperatures may reach up to 90 °C under certain conditions [46], making internal–external heat exchange significant. Consequently, high wall condensation rates can occur locally. The corrosion mechanism is governed by the local equilibrium between condensation droplets and the pipe metal. Condensation droplets may become saturated with FeCO<sub>3</sub> (siderite). At high temperatures and low condensation rates, such saturation allows for the formation of a dense FeCO<sub>3</sub> deposit that substantially reduces the corrosion rate [49,50]. However, the effectiveness of siderite as a protective layer depends on droplet renewal frequency and the overall aggressiveness of the environment. These phenomena are particularly relevant at dew point “cold spots”, characterised by a high water condensation rate (WCR), where pipeline thermal insulation is compromised.

### 3.2.1. Factors Influencing TLC Mechanisms

TLC corrosion phenomena [49] arise under several conditions:

- Pipeline walls separating a hot internal environment from a colder external one (e.g., seawater or soil)
- High internal condensation rates
- High internal temperatures
- High CO<sub>2</sub> partial pressures
- High water-vapour loads.

As previously discussed, the liquid–vapour equilibria of transported mixtures depend on CO<sub>2</sub> composition and contaminant concentrations, potentially inducing unintended phase transitions at temperatures and pressures not usually associated with such phenomena. However, some key parameters that characterise the corrosion rate include gas temperature, CO<sub>2</sub> partial pressure, and condensation rate.

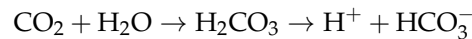
#### Gas Temperature

Gas temperature influences the condensation rate and corrosion-product nature, thus affecting corrosion speed. At low gas temperatures, the condensation and corrosion rate are limited. At higher temperatures, corrosion products may precipitate to form protective deposits that mitigate corrosion. Very low gas temperatures can reach critical condensation rates, but FeCO<sub>3</sub> formation kinetics are unfavourable, leading to supersaturation without protective precipitates and localized attacks that eventually arrest [84]. Around 40 °C, FeCO<sub>3</sub> solubility increases, leading to porous, non-protective siderite layers [49]. Corrosion rates then rise in the initial days and then plateau. If the gas temperature is insufficient to form a coherent protective layer, corrosion occurs uniformly [84]. At high temperatures (70 °C), initial corrosion rates are high due to fast condensation, but decline over time as dense carbonate layers form, although these conditions are not typical of transport scenarios.

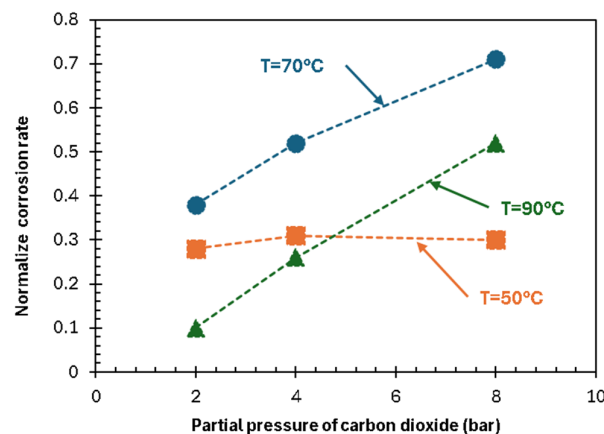
#### CO<sub>2</sub> Partial Pressure

CO<sub>2</sub> partial pressure directly affects corrosion-product scale formation. Low CO<sub>2</sub> partial pressure (0.13 bar) does not favour protective FeCO<sub>3</sub> scabs, resulting in moderate, steady corrosion. In contrast, high partial pressures facilitate protective siderite formation: initial corrosion rates are high, declining over time as exposure increases and FeCO<sub>3</sub> becomes supersaturated. High CO<sub>2</sub> pressure impact is most marked at high condensation rates, where the condensate is hard to saturate and the pH is more affected by CO<sub>2</sub> pressure.

Elevated CO<sub>2</sub> partial pressure increases corrosion by lowering the pH through enhanced hydration and dissociation of CO<sub>2</sub>:



Vitse [46] studied CO<sub>2</sub> pressure effects (1–8 bar) at 50, 70, and 90 °C and varied sub-cooling (difference between fluid and wall temperature). The effect of CO<sub>2</sub> partial pressure is pronounced at high condensation rates (70–90 °C with high sub-cooling), much less so at 50 °C or without sub-cooling (Figure 7). Other studies [85] similarly show that the corrosion rate is visibly influenced by the CO<sub>2</sub> partial pressure when paired with high condensation rate. At low condensation, corrosion-product saturation inhibits ion transfer to/from metal, reducing the CO<sub>2</sub> pressure effect on the corrosion rate.



**Figure 7.** Effect of CO<sub>2</sub> partial pressure (pCO<sub>2</sub>) on the corrosion rate of carbon steel under top of the line corrosion (TLC) conditions, adapted from [50].

#### Water Condensation Rate

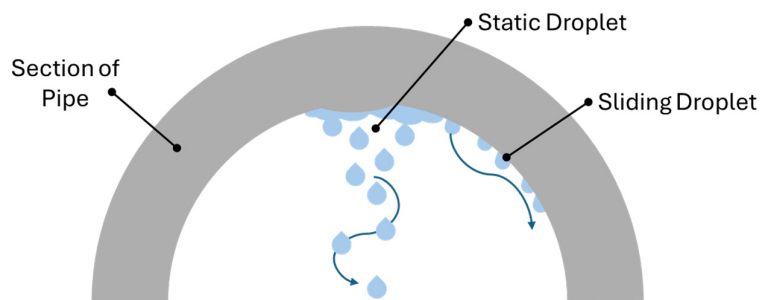
The condensation mechanism occurring within the pipeline walls due to the transport of hot fluid can be described by a characteristic condensation curve that includes a dropwise region, a film wise region, and a transition region [86].

At constant vapour velocity, dropwise condensation occurs at very low surface sub-cooling; this mode may persist even under high sub-cooling and large heat flux conditions. However, when sub-cooling becomes sufficiently large, the condensation rate exceeds the droplet detachment rate, leading to the accumulation of liquid on the surface and the formation of a continuous condensate film (filmwise condensation). Dropwise condensation is a complex phenomenon involving a series of subprocesses that constitute the dynamic “life cycle” of the droplet itself.

Key factors influencing condensation rate under TLC include the following:

- Gas temperature
- Surface sub-cooling
- Incondensable gas concentration
- Gas velocity
- System pressure
- Pipeline inner diameter.

The gas velocity and pipe diameter significantly influence surface wetting kinetics (Figure 8). At low gas velocity or large diameters, droplets may grow large and fall, creating localised acidic droplets that promote localized attack. At higher gas velocity, condensate rivulets slide along the wall forming film-like morphology, leading to more uniform corrosion.



**Figure 8.** Schematic representation of dropwise condensation modes on the upper walls of a pipeline, illustrating static droplets and sliding droplets along the internal surface of the pipe.

To initiate TLC, a critical condensation rate (CCR) between  $0.15$  and  $0.25 \text{ mL m}^{-2} \text{ s}^{-1}$  must be reached in real transport applications [47]. This range serves as an engineering threshold for moist  $\text{CO}_2$  transport, though it may not hold for all real conditions. Generally, condensation rates well above CCR trigger severe corrosion; the pH becomes more sensitive to  $\text{CO}_2$  partial pressure since Fe ions (normally buffering pH) cannot compensate. Conversely, condensation rates below CCR favour  $\text{FeCO}_3$  saturation and protective film formation, raising the pH and slowing the  $\text{CO}_2$  corrosion kinetics. At rates just above CCR, the corrosion rate decreases relative to the critical rate due to film formation [50].

Additionally, gas velocity alters the condensation rate: Vitse and Nešić [46] found that at both high and medium temperatures ( $50$ – $90$  °C), the condensation rate significantly decreases as the gas velocity declines (from  $8$  to  $2$  m/s); the mass transfer from the gas phase is less efficient at low flow turbulence.

### 3.2.2. TLC Mechanisms in $\text{CO}_2$ -Dominant Environments

Initially, condensate droplets on steel surface produce high corrosion rates that stabilise as protective carbonate layers form—similar to sweet corrosion. However, TLC exhibits the following distinctive features:

- $\text{FeCO}_3$  saturation forms a highly protective scale during early stages, limiting Fe ion release and metal dissolution. Yet continuous renewal of condensate (lacking Fe ions) sustains corrosion.
- Equilibrium is reached only when  $\text{FeCO}_3$  saturation approaches unity; the Fe ion input from steel matches the dilution by fresh condensate.
- High condensation rates promote rapid solution renewal at walls, hindering stable protective film development and maintaining high corrosion rates; conversely low condensate mass flux supports protective scale formation, yielding low but non-zero corrosion rates.
- Temperature strongly affects  $\text{FeCO}_3$  deposition kinetics. At low temperatures ( $40$  °C), the corrosion rate is steady but moderate due to inhibited protective film formation. At temperatures  $> 70$  °C, protective film quickly grows but forms surface cracks. Defects in the coating foster localized TLC attacks, which are eventually limited over time: the electrolyte within fissures promotes film growth, partially occluding defects. Localized attack rates can reach  $5$ – $10$  mm/year depending on the condensation rate and environmental aggressiveness.

### 3.3. Role of Other Impurities on Corrosion Behaviour

As previously discussed, the  $\text{CO}_2$  transported in the context of CCTS, due to the various emission sources and the limited capacity for stream purification, may contain impurities such as  $\text{O}_2$ ,  $\text{SO}_x$ , and  $\text{NO}_x$ . These alter the thermodynamic properties of the fluid and thus promote corrosion of steels not only at low partial pressures but also under

supercritical conditions. Some considerations can be made regarding specific impurities by analysing their individual roles in the associated corrosion mechanisms. However, it is important to note that complex mixtures composed of multiple impurities may affect corrosion phenomena differently compared to the effects observed with single species.

### 3.3.1. Oxygen (O<sub>2</sub>)

The effect of O<sub>2</sub> in CO<sub>2</sub> on steel corrosion rates, particularly under s-CO<sub>2</sub> conditions, has been widely investigated in the literature [61,64,65,70,74,87–89], as summarized in Table 6. In general, the corrosion rates of carbon steels in water-saturated CO<sub>2</sub> environments containing O<sub>2</sub> exceed 0.1 mm/year.

**Table 6.** Comparison of the effect of O<sub>2</sub> on the corrosion rate of steels in s-CO<sub>2</sub>.

N°	P (bar)	T (°C)	H <sub>2</sub> O (ppmv)	O <sub>2</sub> (ppmv)	Steel	t (h)	Flow (rpm)	Corrosion Rate (mm/anno)	Refs.
1	75.8	40	2440	100		5	Static		[74]
2	80	50	Saturated (10 g)	0 2% (1.6 bar) 4% (3.3 bar) 6% (5.1 bar)	X65	24	static	0.38 0.6 1 0.9	[65]
3	80	50	650 2000 3000	3.3 bar 4%	X65	24	Static	NO corrosion NO corrosion ≤0.01	[64]
4	79.6–82	35	Saturated (100 g)	0	304 L 316 L X42 X60	120	100	0.002 0.001 0.014	[61]
5	94.8–103	49	Saturated (100 g)	3 v%	304 L 316 L X42 X60	120	100	0.003 0.004 0.099 0.093	[61]
6	100	60	Saturated (1 mL 55.6 mmol)	Yes (–1000 ppm)	X42	120	Static	0.008 (–1 mg) No corrosion	[89] [89]
7	100	20	1220	488	x65	720	Static	No corrosion	[70]
8	100	10	50 v% 20 (Saturated) 50 50 10 20 50	0 0 0 200 100 200	X65	312 0.336 336 336 312 336 432	Static	0.5 0.8 0.5 2.7 1.2 1.3 0.6 (pit corrosion rate 17)	[70]
9	150	80	Saturated	1000 ppm		288	120	0.2–0.9	[90]
10	150	100	5% NaCl	0.045 bar 0.45 bar	Type 420	720	static	0.08 Localized 0.25 Localized	[88]
11	300	100	5% NaCl	0.045 bar 0.45 bar	Type 420	720	static	0.07 Negligible localized 0.34 Localized	[88]
12	150	100	5% NaCl	0.0045 bar 0.45 bar	M-SS	720	Static	0.01 localized 0.04 Localized	[88]
13	300	100	5% NaCl	0.045 bar 0.45 bar	M-SS	720	Statie	0.02 No Localized 0.07 localized	[88]
14	100		12 13	1000 0	X65	48–96	0–3 m/s	5 to10 3 to 5	[70]
15	80	50	400 ml Water phase	4%; 0	X65	24 120	Static	19.3; 19.2 14.1; 10.6	[76]
16	80		50,400 mL Water phase	4%	3Cr		Static	0.01	[64,65]

Small amounts of oxygen (100 ppm) in experiments conducted at 7.58 MPa and 40 °C with 1000 ppm of water in undersaturated s-CO<sub>2</sub> showed no significant effect on the corrosion rate of carbon steels [74]. Similar results were obtained even at higher O<sub>2</sub> concentrations, as long as the water content remained below its solubility limit in s-CO<sub>2</sub>

under the test conditions [64,70]. However, in water-saturated s-CO<sub>2</sub> environments, the addition of O<sub>2</sub> can lead to severe corrosion.

Studies on the effect of different O<sub>2</sub> contents (0%, 2%, 4%, and 6%) on the corrosion of X65 steel in water-saturated CO<sub>2</sub> at 8 MPa and 50 °C after 24 h of exposure showed an increase in corrosion rate, reaching a maximum value of 1 mm/year with 4% O<sub>2</sub> addition [65,76].

In the absence of O<sub>2</sub>, the steel surface is covered with a dense, protective siderite (FeCO<sub>3</sub>) layer. When O<sub>2</sub> is present in the system, it can inhibit FeCO<sub>3</sub> formation and promote the formation of porous iron oxides with reduced protective capability, resulting in increased corrosion rates. Consequently, in water-saturated CO<sub>2</sub> environments containing O<sub>2</sub>, carbon steels (e.g., X42 and X60) become more susceptible to corrosion [61].

When varying the temperature in dense-phase CO<sub>2</sub> environments containing 50 vol% water at 10 MPa with different O<sub>2</sub> concentrations (0, 100, and 200 ppm), it was observed that the addition of O<sub>2</sub> increases the corrosion rate of carbon steel X65 by 50–120% [66]. At 50 °C, Fe<sup>2+</sup> reacts with O<sub>2</sub>, forming iron oxides, which results in a lower Fe<sup>2+</sup> concentration in the dense CO<sub>2</sub> phase compared to that without O<sub>2</sub>. The reduced Fe<sup>2+</sup> content destabilizes the protective siderite layer and leads to local failure of the FeCO<sub>3</sub> film. As a result, severe localized corrosion can occur with rates up to 17 mm/year.

An important aspect to consider is the exposure time, which significantly affects the evaluation of oxygen's influence on corrosion rates. For example, in tests with carbon steels exposed to water-saturated s-CO<sub>2</sub> at 50 °C and 8 MPa, it was found that during the initial corrosion period the addition of 4% O<sub>2</sub> had little to no effect, with corrosion rates of 19.2 mm/year without O<sub>2</sub> and 19.3 mm/year with 4% O<sub>2</sub>. However, when the immersion time was extended to 120 h, the corrosion rate with 4% O<sub>2</sub> (14.1 mm/year) was notably higher than that without O<sub>2</sub> (10.6 mm/year) [65,76].

### 3.3.2. Nitrogen Dioxide (NO<sub>2</sub>) and Sulphur Dioxide (SO<sub>2</sub>)

The effects of equal concentrations of O<sub>2</sub>, SO<sub>2</sub>, and NO<sub>2</sub> on the corrosion of carbon steel in the s-CO<sub>2</sub>/H<sub>2</sub>O system at 40 °C and 7.58 MPa have been compared [69,70,74,77]. It was observed that 100 ppm of O<sub>2</sub> had no effect on the corrosion rate of carbon steel, whereas the corrosion rate increased from 2.3 to 4.6 mm/year in the presence of 100 ppm of SO<sub>2</sub>. However, with 100 ppm of NO<sub>2</sub>, the maximum corrosion rate of carbon steel increased significantly, reaching up to 11.6 mm/year.

The effect of NO<sub>2</sub> can lead to significantly higher corrosion rates of carbon steel (0.06–1.6 mm/year [76]) compared to the same concentration of SO<sub>2</sub> (≤0.02 mm/year). In particular, the corrosion rate decreases with decreasing NO<sub>2</sub> content but remains higher in the s-CO<sub>2</sub>/NO<sub>2</sub>/H<sub>2</sub>O environment. This suggests that NO<sub>2</sub> dissolves in water to form HNO<sub>3</sub>, leading to a sharp decrease in pH. HNO<sub>3</sub> exhibits a strong oxidizing effect on Fe<sup>2+</sup> ions, resulting in the formation of a powdery, rust-like corrosion product. This corrosion film is soft and less protective, thus significantly increasing the corrosion rate of carbon steel in the presence of NO<sub>2</sub> [67]. Therefore, in a CO<sub>2</sub> system containing water, the following apply:

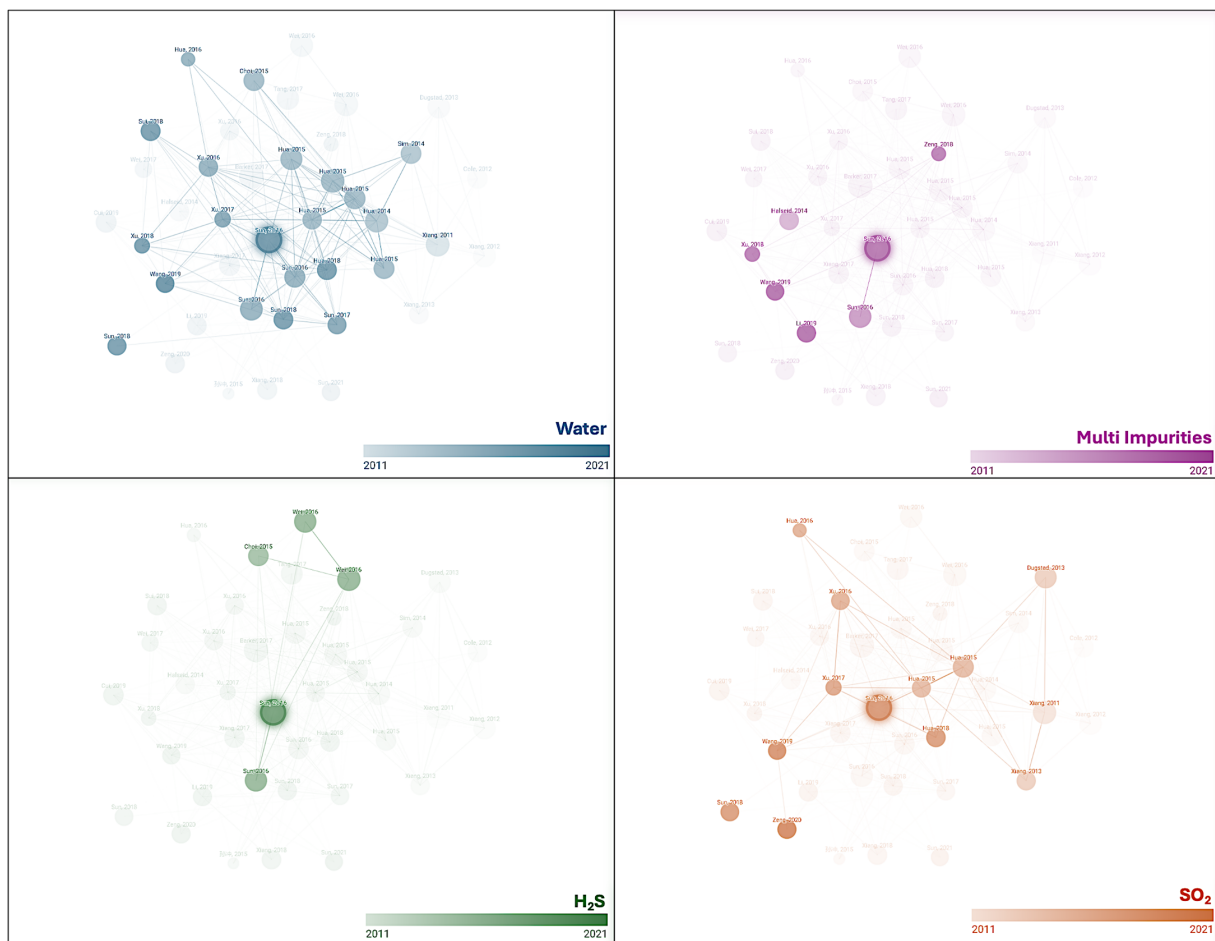
- O<sub>2</sub>, SO<sub>2</sub>, and NO<sub>2</sub> can accelerate the corrosion of carbon steels in CO<sub>2</sub>/H<sub>2</sub>O environments. Among the three impurities at the same concentration, NO<sub>2</sub> has the most significant effect on the corrosion rate of carbon steel, followed by SO<sub>2</sub> and O<sub>2</sub>
- For SO<sub>2</sub>, corrosion can occur in the CO<sub>2</sub>/SO<sub>2</sub>/H<sub>2</sub>O system even when the water content is well below the solubility limit of CO<sub>2</sub> in water. The corrosion rate increases with the concentration of SO<sub>2</sub>.

### 3.3.3. Effects of Real Impurity Mixtures

#### Effect of Mixed Impurities

The corrosion behaviour associated with mixtures of multiple impurities under various pressure and temperature conditions may differ significantly from that estimated in previous cases with single-species environments. This deviation arises from interactions among the different species in the environment. Several research groups worldwide have published systematic studies with extensive datasets on corrosion, including Choi et al. [63,75,76,89], Xiang et al. [91,92], Hua et al. [93–95], Xu et al. [96,97], Sun et al. [98–100], Yevtushenko et al. [101], and Wei et al. [32,102].

For instance, Figure 9 presents a map of corrosion research data in s-CO<sub>2</sub> environments, highlighting the most studied environments with the presence of water and different impurities. The effect of water on the corrosion rate has been extensively studied, as previously discussed, while the influence of SO<sub>2</sub> and H<sub>2</sub>S has also been investigated. There remains a need to examine the combined effect of multiple impurities on corrosion behaviour. In recent years, a renewed interest has emerged in understanding how complex impurity interactions influence corrosion processes, reflecting the increasingly realistic conditions found in industrial and environmental settings. Despite these extensive efforts, no solid conclusions have yet been reached due to conflicting results reported by different groups. Factors such as differing exposure durations and cleaning methods have influenced the estimation of the corrosion rates.



**Figure 9.** Literature data illustrating the most commonly studied environments for steels in supercritical CO<sub>2</sub>, including the presence of water, H<sub>2</sub>S, SO<sub>2</sub>, and impurity mixtures. The dataset is based on a comprehensive review of the most recent literature, supported by AI-assisted tools to enhance data collection and analysis.

Some studies have attempted to simulate realistic CO<sub>2</sub> transport conditions by adding small amounts of impurities to the main CO<sub>2</sub> stream and testing their effects on pipeline steels. For example, considering a flow of 4 L/h of s-CO<sub>2</sub> containing 1000 ppmv H<sub>2</sub>O (decreasing to 500 ppmv, undersaturated) at 10 MPa and 60 °C, with additional O<sub>2</sub> (8100 ppmv), SO<sub>2</sub> (70 ppmv), NO<sub>2</sub> (100 ppmv), and CO (750 ppmv) [103], it was observed that after 168 h of exposure the generalized corrosion rate for carbon steel (X52) was approximately 0.00003 mm/year. In contrast, pitting corrosion was observed for a CrMo alloy (UNS G41400) [103,104], which could be mitigated by appropriate heat treatment of the material [105].

For carbon steels X52 (L360NB) and X70 (L485MB) under s-CO<sub>2</sub> conditions (11 MPa and 60 °C), the addition of O<sub>2</sub> increased the corrosion rate [106]. The corrosion rate for X70 decreased significantly with a reduction in O<sub>2</sub> content, while for X52, O<sub>2</sub> concentration had no noticeable effect. Nonetheless, the corrosion rates for both carbon steels remained below 0.1 mm/year, indicating that a small amount of water in s-CO<sub>2</sub> (undersaturation) has limited impact on corrosion. Additionally, adding a small amount of CO (50 ppmv) or SO<sub>2</sub> (70 ppmv) to the mixture did not change the corrosion behaviour. However, the addition of 100 ppmv NO initiated corrosive phenomena [106]. The presence of multiple impurities significantly increases the average corrosion rate, up to twice as high in some cases. The complex synergistic interactions among CO<sub>2</sub>, O<sub>2</sub>, H<sub>2</sub>S, and SO<sub>2</sub> contribute to the highest observed corrosion rates [107,108].

#### Amine Treatment and Effects on Corrosion Rates

The most widely adopted method for CO<sub>2</sub> separation from flue gases is the amine-based process [109], favoured due to the relatively low cost of ammonia used as a sorbent and its high CO<sub>2</sub> loading capacity. This has prompted the development of numerous studies aimed at improving its industrial application [110]. However, it is possible for the CO<sub>2</sub> stream downstream of treatment to be contaminated with residual amine or ammonia.

Studies have investigated the effects of amine residues (100 ppm water + 100 and 1000 ppm of amine) on the water solubility limit in s-CO<sub>2</sub>, the pH value of the condensed aqueous phase, and the corrosion rate of carbon steels [77]. Amines significantly increase the pH of the condensed water, thereby reducing the corrosion rate of carbon steel: for 100 ppm of amine, corrosion rates were estimated at 0.1 mm/year, and for 1000 ppm at 0.01 mm/year.

Moreover, the addition of 50 wt% ethylene glycol (MEG) in a NaCl solution saturated with CO<sub>2</sub> at 8 MPa and 50 °C decreased the corrosion rate of X65 steel from 4.6 to 1.7 mm/year [32].

## 4. Conclusions

It is important to consider the supply chain and origin of flue gases as their composition is strongly dependent on several operational factors, including flame temperature, fuel type, and process conditions. In addition, flow rates and the stability of both mass flow and gas composition may vary significantly, further influencing downstream behaviour.

It can be concluded that, in the absence of water, CO<sub>2</sub> (whether in supercritical, gaseous, or liquid form) does not generally induce significant corrosion under low-temperature conditions or result in negligible corrosion rates. However, in the presence of water, the CO<sub>2</sub>-induced corrosion mechanism becomes significantly more complex and depends on several interacting factors. These include the partial pressure of CO<sub>2</sub>, which in the absence of other species governs the pH of the aqueous phase, as well as the pH itself, which can be modified by the presence of other chemical species affecting equilibrium conditions through changes in CO<sub>2</sub> solubility. Temperature also plays a key role, as it

influences both the solubility of CO<sub>2</sub> in the aqueous phase and the nature, morphology, and stability of the corrosion-product scales. Furthermore, the presence of additional species can alter cathodic reaction kinetics, modify pH, and influence the composition and protectiveness of the corrosion scales. Flow rate is another critical parameter, as it affects both the formation and the mechanical stability of the corrosion layers.

Under these conditions, several predictive models for the corrosion rate of carbon steel are available. In addition, more advanced approaches explicitly consider corrosion-product formation and its coupling with microstructural evolution. However, the development of reliable predictive models remains challenging due to the complexity associated with CO<sub>2</sub> phase behaviour and phase transitions under transport conditions.

In wet gas environments, water condensation is a key factor governing corrosion severity. In CO<sub>2</sub>–water systems, the role of impurities becomes particularly significant. For instance, small amounts of oxygen (approximately 100 ppm) under experimental conditions of 7.58 MPa and 40 °C with 1000 ppm water in supercritical CO<sub>2</sub> (undersaturated conditions) have been shown to have no substantial effect on the corrosion rate of carbon steels. In contrast, O<sub>2</sub>, SO<sub>2</sub>, and NO<sub>2</sub> can accelerate corrosion in CO<sub>2</sub>/H<sub>2</sub>O systems, with NO<sub>2</sub> exhibiting the strongest effect, followed by SO<sub>2</sub> and O<sub>2</sub> when compared at similar concentrations. For SO<sub>2</sub>, in particular, corrosion can occur even when water content is below the solubility limit of CO<sub>2</sub> in water, and the corrosion rate increases with increasing SO<sub>2</sub> concentration.

The presence of impurities also alters the gas–liquid equilibrium boundaries compared to pure CO<sub>2</sub> systems. Acidic impurities generally enhance corrosion in s-CO<sub>2</sub> environments when water is present, with the exception of H<sub>2</sub>SO<sub>4</sub>. The addition of HNO<sub>3</sub> may further promote localized corrosion phenomena such as pitting. On the other hand, alkaline species such as amines and NaOH tend to reduce the corrosion rate of carbon steels. Amines, in particular, significantly increase the pH of condensed water, leading to reduced corrosion rates, with reported values decreasing from approximately 0.1 mm/year at 100 ppm amine to as low as 0.01 mm/year at 1000 ppm. Additionally saline impurities such as Cl<sup>−</sup>, NO<sub>3</sub><sup>−</sup>, and SO<sub>4</sub><sup>2−</sup> can significantly increase corrosion rates in s-CO<sub>2</sub> environments and may also promote localized corrosion mechanisms.

Future research should focus on validating existing predictive models through dedicated in-field or industrial-scale experiments, as laboratory data alone remain insufficient to fully capture real operating conditions. In particular, the development of reliable predictive approaches is hindered by the complexity of CO<sub>2</sub> phase behaviour and phase transitions under pipeline conditions. A more detailed understanding of how phase diagrams are modified in the presence of impurities would be highly beneficial, as this could allow the identification of key, directly measurable parameters that can serve as practical indicators for corrosion risk assessment in real systems.

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