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4 Shrinkage and creep of high-performance concrete based on 5 calcium sulfoaluminate cement

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17 **ABSTRACT**

18 Calcium sulfoaluminate (CSA) cements can be used in concrete as an alternative binder, as full or
19 partial replacement of Portland cement. This enables reducing CO₂ emissions from cement production
20 and offers other advantages, e.g. fast gain of mechanical properties. However, very little data is
21 available on the behaviour of concretes with CSA binders, in particular on their creep and shrinkage.
22 This paper presents a study on volume stability of high-performance concretes with CSA binder,
23 blended with, or completely replacing, Portland-limestone cement. Shrinkage and creep in both
24 autogenous and drying conditions were measured from 1 day until 1 year. The experimental results
25 were compared with the empirical models Model Code 2010 and ACI 209.R-92. The results indicate
26 that standard models originally developed for Portland cement concretes do not allow to address the
27 pronounced differences between the Portland and CSA-based concretes, since their main input data
28 are merely based on compressive strength.
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32 **Keywords:** Calcium sulfoaluminate cement, high performance concrete, shrinkage, creep, model
33 code.
34

35 **1. INTRODUCTION**

36 Calcium sulfoaluminate cement (CSA) represents an ecological alternative to Portland cement (PC)
37 due to its reduced CO₂ emissions during production [1]. The main constituent of CSA clinker is
38 ye'elimite (C₄A₃ \bar{S}), in combination with other phases like belite (C₂S), ferrite (C₄AF), mayenite
39 (C₁₂A₇) and anhydrite, depending on the raw materials used [2,3]. Usually, around 20% by mass of
40 calcium sulphate is added to CSA clinker to regulate the setting time, the strength development and
41 the volume stability of the cementitious matrix [2], creating a binary system. The main hydration
42 product is ettringite (AFt), which is formed from the reaction of ye'elimite, lime and sulphates, the
43 latter of great importance for their kinetics [4]. A possible alternative to the pure system is represented
44 by blends of CSA and PC. In these blends, the hydration process is governed by the calcium
45 sulfoaluminate cement in the first days, while the Portland cement reacts later [5]. The precipitation
46 of AFt during early-age hydration is mainly responsible for the development of the material
47 properties, like fast setting, rapid hardening [6] or expansion for shrinkage compensation [6,7]. After
48 the rapid CSA reaction in the first days, the PC starts to react, leading to long-term performance
49 improvement [8].

50 In the 1970s, the People's Republic of China started a large research program on calcium
51 sulfoaluminate and calcium sulfoaluminate belite cements aiming to improve the knowledge about
52 this class of binders. Twenty years of experience in production (albeit with low amounts of CSA
53 produced yearly) and use led to standardization both of the cement and of concrete mixture
54 compositions [9]. The main outcome of this program was the need to focus on the rapid-hardening
55 behavior of these materials, which is its most useful and powerful characteristic, though it is also the
56 most complex aspect to control. The Chinese experience is the only example so far of a research
57 program of such dimensions and complexity on CSA. In recent years, significant steps forward in the
58 basic understanding of CSA's chemical and physical properties were accomplished. However, the
59 practical use of CSA is still limited to niche products, e.g., as expansive agent to compensate
60 shrinkage of PC concrete [10–12], in the production of self-levelling screeds [13], as sealing mortar
61 for road works [14] and as tile adhesive [15].

62 While the dimensional stability of calcium sulfoaluminate systems under water cured conditions has
63 been studied (e.g., [16]), its application in structural concrete with other exposure conditions of the
64 surface should be examined. In particular, sealed conditions in the first days of hardening
65 (representative for the period before demolding) followed by drying conditions at later ages (after
66 demolding) [17] need to be investigated. Due to the initial fast reaction of CSA systems, particular
67 attention is required in the first stage, when the material is under autogenous conditions [18]. This is
68 of great importance especially when the water-to-cement ratio (w/c) is lower than the minimum value
69 required by the stoichiometry of the cement hydration reaction [19]. This value represents the limit
70 between systems that may theoretically achieve full hydration and systems where a portion of the

71 cement remains unreacted due to insufficient amount of water for hydration, causing self-desiccation.
72 This w/c is around 0.40-0.42 for PC and depending on the amount of added calcium sulphate up to
73 0.78 for pure ye'elinite [3]. For a specific CSA, a value of 0.75 was recently measured [20].
74 Considerable pore fluid pressures are generated in a system that undergoes self-desiccation, leading
75 to autogenous deformation and build-up of self-induced stresses that must be limited to avoid
76 cracking [21,22]. At the same time, creep and relaxation play an important role in both deformation
77 and cracking behavior of the new systems; to the authors' best knowledge, very limited data has been
78 published so far on this aspect for the CSA-based systems.

79 Building up on a previous publication that focused mainly on the mechanical properties [23,24], this
80 study has the aim of investigating the deformation behavior of concrete systems containing either
81 pure CSA or a blend with PC, in terms of shrinkage and creep evolution in both autogenous and
82 drying conditions. This study covers the stage after concrete demolding and up to 1 year of age,
83 underlining the possible interactions between the two investigated conditions on the stability of the
84 matrix, with and without external mechanical loads. The experimental results are compared to
85 shrinkage and creep predictions according to Model Code 2010 [25] and ACI 209.R-92 (1992) [26]
86 to highlight the differences with PC-based concrete and assess whether new predictive models need
87 to be developed for these systems.

88

89 **2. MATERIALS AND METHODS**

90 **2.1. Materials**

91 Three concrete mixtures made with 1) pure CSA, 2) a Portland-limestone cement, CEM II A-LL 42.5
92 R, and 3) a blend of the two (50/50 ratio by mass) were considered for this study; in the following,
93 they will be labelled as CCSA, CPC and CMIX, respectively. The CSA had as main components (by
94 mass %, measured by XRD): ye'elinite 53.0, anhydrite 18.7, bredigite 10.7, C₂S 7.7. The Portland-
95 limestone cement was chosen for the stabilization effect of the calcium carbonate on the ettringite
96 that will form during hydration [15,27,28]. Its phase composition (by mass %, measured by XRD)
97 was: C₃S 55.4, C₂S 9.3, C₃A 3.3, C₄AF 9.7 and calcite 12.4.

98 The mix design is based on the mixtures developed for a previous study [8], which were designed
99 with the target of reaching the same 28-day strength class C50/60 and at least 160 mm slump after 45
100 minutes from water addition. This approach was chosen because it would be the practical approach
101 followed by a concrete producer. In order to reach the same strength class, the w/c and the amount of
102 cement in the three mixtures were different (see Table 1), while the total amount of aggregates was
103 similar (small adjustments were however necessary for workability reasons). The paste volume was
104 almost identical in the three mixtures. The mixtures were prepared with siliceous coarse rounded
105 aggregate (maximum diameter 20 mm) and river sand conforming to the EN 12620 standard. A

106 polycarboxylate-based superplasticizer (SP) in liquid form was used in all concretes. Compared to
 107 the original mix design used in [8], the superplasticizer amount was lower thanks to the more efficient
 108 concrete mixer (see below) used during this project. Citric acid as hydration retarder and lithium
 109 carbonate as set inductor, both in powder form, were added to the concretes containing CSA cement.
 110 A CEM II A-LL 42.5 R (according to EN197-1) and a commercial CSA cement (i.tech ALI CEM
 111 GREEN® by Italcementi) were used. The mixtures proportions are reported in Table 1.

112
 113 *Table 1. Concrete mixtures design [kg/m³].*

	CPC	CCSA	CMIX
CEM II-LL 42.5R	450	-	200
CSA ALI CEM	-	350	200
Sand 0.20-0.35	258	265	255
Sand 0.6-1.0	197	202	195
Sand 1.5-2.5	262	270	260
Gravel 3-4	142	146	141
Gravel 6-10	335	344	331
Gravel 10-20	584	600	578
SP (% on binder)	1.0	0.3	0.5
Retarder (% on binder)	-	0.4	0.5
Accelerator (% on binder)	-	0.1	0.2
Water	157.5	175	180
w/c	0.35	0.50	0.45

114

115 **2.2. Mixing procedure**

116 Before mixing, all the aggregates were dried at 50% RH. The additional amount of water required to
 117 obtain saturated surface dry (SSD) conditions was taken into account in the mix design; an absorption
 118 of around 1% by mass in SSD condition was considered for all aggregates. An Eirich R08W high-
 119 shear mixer was used. Firstly, all the dry aggregates were placed into the mixer with 1/3 of the total
 120 water, followed by mixing for 10 s and a 2 min period of rest to complete the absorption. Afterwards,
 121 cement was added and the mixing procedure started, adding gradually the rest of the water and letting
 122 it homogenize for 1 min. The admixtures were added thereafter, starting with the superplasticizer
 123 followed by the two powders, mixing for 1 min after each addition; the initial consistency was
 124 controlled after one more final minute mixing: if it resulted homogeneous, the mixing operation was
 125 concluded; otherwise, 1 min more was added to the final step. The total time required by the whole
 126 mixing operation was about 5-6 min.

127

128 **2.3. Fresh state properties**

129 Some investigations on the fresh state were performed as quality control measure for each mixture.

130 The Abrams cone was used to measure the workability after mixing (around 7-8 minutes after water

131 addition) by measuring the slump according to the EN 12350-2 standard. The air content and the
132 density were measured following the EN 12350-6 and EN 12350-7 standards, respectively.

133

134 **2.4. Compressive strength**

135 The measurements of compressive strength were performed at 1, 7 and 28 days following the EN
136 12390-3 standard on duplicate cubic samples (edge 150 mm) for each age of testing. The samples
137 were produced by filling the plastic molds in two layers and vibrating for 10 s after each pouring.
138 Once the upper surfaces were rectified, the samples in the molds were stored in a climatic room at
139 $20\pm 0.5^{\circ}\text{C}$ and $\text{RH} > 95\%$ covered with plastic sheets. The samples were demolded at 24 h and stored
140 in the same climatic room until testing. Before testing, the cubes were polished on the two loading
141 surfaces.

142

143 **2.5. Shrinkage and creep**

144 The volume changes with and without load (creep and shrinkage) were measured both under
145 autogenous and drying conditions following the Swiss standard SIA 262/1:2013 F until 364 days. In
146 particular for creep, CMIX was monitored up to 1 year, while the measurements on the other mixtures
147 were stopped after 182 days because of their substantially stable behaviour, especially when
148 compared to CMIX. It was also decided to deviate from SIA 262/1:2013 F in terms of environmental
149 RH, which was 57% instead of 70% for both shrinkage and creep samples and loading steps applied
150 at different ages. These changes were adopted to observe a higher deformation response (at lower
151 RH) and upgrade the applied stress following the strength evolution.

152 The specimens were $120\times 120\times 360\text{ mm}^3$ prisms cast in stainless steel molds, covered with plastic
153 sheets and stored in the first day after casting in a climatic room at $20\pm 0.5^{\circ}\text{C}$ and $\text{RH} > 95\%$. The
154 samples were demolded after 24 h, when the test started. Two samples were prepared for each test
155 and condition. For every sample, two measuring pins were glued at a distance of $250\pm 1\text{ mm}$ on two
156 opposite sides of the sample, avoiding the cast surface. In this way, two length measurements on each
157 of two samples could be performed. Considering the drying condition, all the surfaces of the sample
158 were exposed to a $20\pm 0.5^{\circ}\text{C}$ and $57\pm 3\%$ RH environment, while sealed conditions were assured by
159 covering completely the samples with adhesive aluminum tape just after demolding in order to avoid
160 moisture loss. The shrinkage samples were stored horizontally on two thin blade supports, to allow
161 drying from all their surfaces. The duplicate creep samples were placed in hydraulic creep stations
162 vertically one over the other with a metallic plate in between. In this way, the same load could be
163 applied on the duplicate samples. Three sequential loading steps at 1, 7 and 28 days were applied to
164 the samples in order to investigate the initial creep evolution, which is particularly important for the
165 rapid-hardening behavior of CSA cement, and the evolution of creep at increasing load. The first load

166 step took place after 1 day, the second after 7 days and the last after 28 days, such that the total applied
 167 stress σ corresponded to 1/3 of the compressive strength at the age of loading (see Fig. 3). However,
 168 because of the limited load capacity of the creep testing setups, some samples were loaded at 1/4 of
 169 the compressive strength only. These levels of stress were chosen aiming at linear creep behaviour
 170 [29].

171 The deformations were measured manually at selected ages with a comparator equipped with a digital
 172 deformation transducer (resolution 0.001 mm corresponding to 4 $\mu\text{m/m}$ for the 250 mm measuring
 173 base). The shrinkage and creep results were expressed in terms of strain ε [$\mu\text{m/m}$]; the creep data were
 174 obtained by subtracting from the total strains the strains due to instantaneous deformation (assumed
 175 as occurring between immediately before loading and about 5 min after loading) and subtracting the
 176 shrinkage strain measured on the companion shrinkage samples.

177 All data shown in the graphs in the results section represents the average of two samples and is plotted
 178 with its standard deviation.

179

180 3. RESULTS

181 3.1. Fresh state

182 A summary of the results on fresh concrete is reported in Table 2. Considering CPC and CMIX, the
 183 density was slightly lower than expected at the mix design stage, when a value of 1.5% of entrained
 184 air was considered; the corresponding differences were also found for the air content. In terms of
 185 slump, high initial workability for the CSA based mixtures was found, with no aggregate segregation.

186

187

Table 2. Details of the fresh state tests

		CPC	CCSA	CMIX
Initial slump* [mm]		180	250	225
Density [kg/m ³]	measured	2368	2355	2309
	mix design	2392	2352	2346
Air content [%]		2.5	1.3	3.2

188 *Measured at the end of casting operations, 7-8 min after water addition

189

190 3.2. Compressive strength

191 The compressive strength evolution was monitored until 28 days; the results are reported in Figure 1.
 192 Compared to the previous study on the same mixture compositions [23], these mixtures reached lower
 193 compressive strength at 28 days due to the higher amount of entrained air, probably connected to the
 194 different mixer used in this campaign.

195

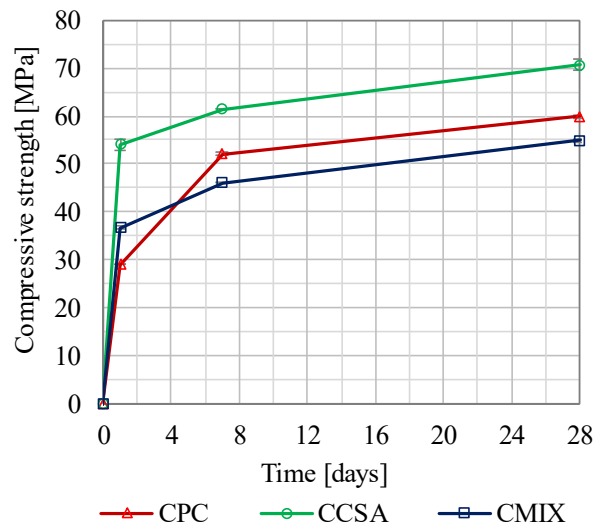


Figure 1. Compressive strength results. Average and standard deviation of 2 samples.

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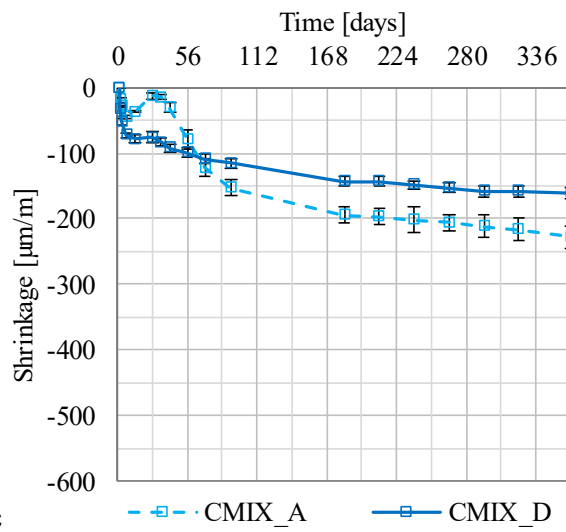
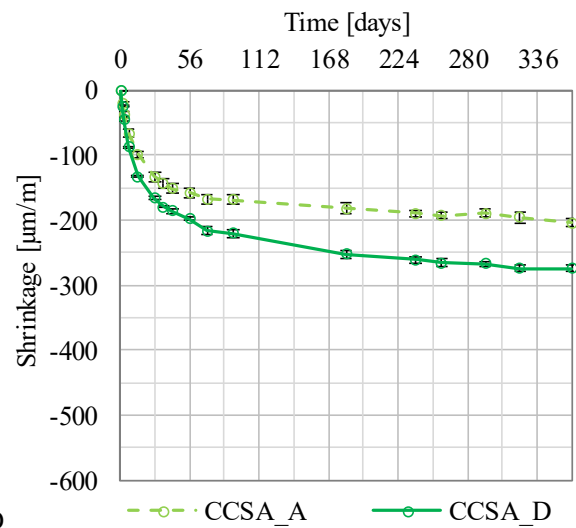
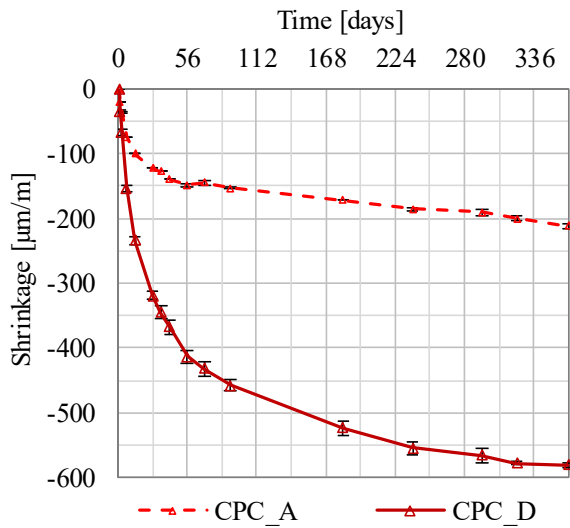
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199 3.3. Shrinkage

200 Results of shrinkage under both autogenous and drying conditions are reported in Figure 2. The dotted
 201 lines indicate the sealed condition, while continuous lines indicate samples exposed to the
 202 environment of the climatic room (20°C and 57% RH). In drying conditions, CPC showed the fastest
 203 evolution and the highest final value of shrinkage. At the same time, in autogenous conditions the
 204 final shrinkage at 1 year was similar for all the mixtures (-212, -204, -228 $\mu\text{m}/\text{m}$ for CPC, CCSA and
 205 CMIX, respectively). While CPC and CCSA followed the same trend, CMIX showed a particular
 206 evolution in time: initial shrinkage was observed up to 7 days, followed by expansion until 28 days
 207 and a more stable period up to 35 days. From that point on, CMIX started to shrink again: first rapidly
 208 until 56 days, then at a slower steady rate. During this second shrinking period, the curve for
 209 autogenous conditions crosses the shrinkage curve in drying conditions (the difference remains
 210 however small, about 50 $\mu\text{m}/\text{m}$).

211 The overall result is lower total shrinkage for CCSA and CMIX compared to CPC, especially in
 212 drying conditions.

213



214

a

b

c

215

216 *Figure 2. Autogenous (_A) and drying (_D) shrinkage results. Average and standard deviation of 2 samples.*

217

218 3.4. Creep

219 Figure 3 shows the creep evolution for the three mixtures. The results under autogenous and drying
 220 conditions are reported, corresponding to basic and drying creep, respectively. When one considers
 221 different values of the applied stress (note in particular considerably lower applied stress at 1 d for
 222 the CPC than for the other concretes), it becomes evident that the specific creep (i.e. creep per applied
 223 stress) was considerably higher for the CPC compared to CCSA and CMIX, in particular in drying
 224 conditions. Furthermore, the difference between the drying and basic creep was the highest for the
 225 CPC system. An interesting trend could be observed for the blended system, CMIX, where the basic
 226 creep became larger than the drying creep after about 42 d; similar behaviour of larger deformations
 227 in sealed than in drying conditions could be observed for shrinkage results in Fig. 2c.

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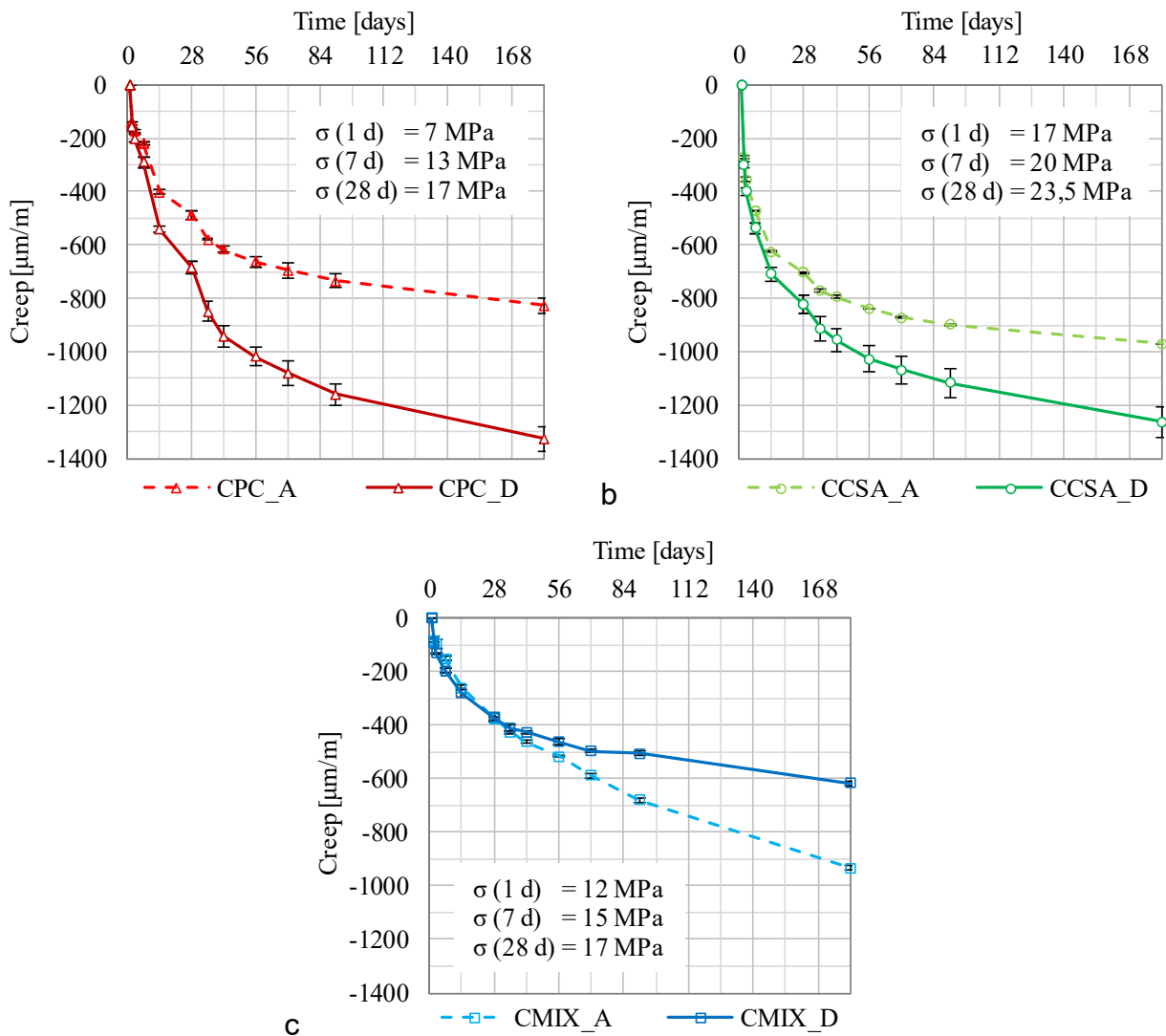


Figure 3. Basic (_A) and drying (_D) creep results. Stresses applied at 1, 3 and 28 d are indicated.

4. DISCUSSION

4.1. Differences between autogenous and drying conditions

The measurements of shrinkage and creep both in sealed conditions (autogenous shrinkage and basic creep) and in drying conditions (drying shrinkage and drying creep) may be considered representative of the boundaries of the concrete behaviour in the field (when one disregards the inevitable temperature changes). Sealed conditions are characteristic of the concrete before demolding, when no moisture exchange between the material and the outer environment takes place, while drying conditions occur after demolding. Considering then concrete elements of large dimensions, the core can be considered to hydrate in sealed conditions, while the surface undergoes drying.

Before examining the volume changes of the concrete mixtures, it must be pointed out that the microstructure and the pore structure of the matrix depend strongly on the binder used. The PC-based systems contain mainly calcium-silicate hydrate (C-S-H), whose creep and shrinkage and their strong dependence on RH are well known [30]. On the other hand, the main hydrated phase of CSA is

246 ettringite, with morphology and porosity radically different from C-S-H. Ettringite is expected to be
247 dimensionally quite stable and less sensitive to changes in RH [31,32]. Hence, different shrinkage
248 and creep behaviours are expected from concrete made with PC and CSA binders.

249 The PC mixture showed the highest difference between autogenous and drying shrinkage. This is
250 generally observed in concrete based on Portland cement, with the relative importance of autogenous
251 shrinkage compared to drying shrinkage growing with the decrease of w/c [19].

252 On the contrary, in CCSA the self-desiccation shrinkage is very close to the total shrinkage measured
253 at 57% RH. This may be due to the combination of rapid self-desiccation and a very tight pore
254 structure that limits further water loss in drying conditions [33]. While the autogenous shrinkage of
255 CPC and CCSA are similar, the total shrinkage in drying conditions is much lower in CCSA.

256 A radically different picture is given by CMIX: in sealed conditions, after an expansion phase [34,35]
257 that ends at about 28 days, autogenous shrinkage followed at high rate and surpassed the total
258 shrinkage in drying conditions after 70 days. The observed trend was different with respect to PC,
259 since the total shrinkage in drying conditions is usually found to be higher than autogenous shrinkage,
260 when the environmental RH is lower than the internal RH in sealed conditions. According to the
261 results obtained on mortars with the same binders as used here for concrete, the internal RH in
262 autogenous conditions after 56 days is around 86% and appears to be stable [36]. This value is not
263 expected to drop much further, since Portland cement hydration will not proceed when the RH is
264 below about 70-80% [37–40] and the hydration of the CSA should be finished by that time. Hence,
265 the internal RH in autogenous conditions should be much higher than the equilibrium RH of the
266 drying specimens (57%).

267 Both drying and autogenous shrinkage can be described by the Biot-Bishop equation [41], a classical
268 poromechanics approach that approximates the linear elastic deformation of an unsaturated porous
269 body under the action of pore pressure:

$$270 \quad \varepsilon_v = -S_w \cdot p \cdot (1/K_b - 1/K_s) \quad (1)$$

271 where S_w (-) is the volumetric saturation degree, p (Pa) is the pore pressure, K_b (Pa) is the drained
272 bulk modulus of the porous body and K_s (Pa) is the bulk modulus of its solid skeleton. The pore
273 pressure can be calculated with the Kelvin-Laplace equation (neglecting the effect of the ions in the
274 pore solution on the internal RH [42–44]):

$$275 \quad p = \rho \cdot R \cdot T \cdot \ln(RH)/M \quad (2)$$

276 where ρ (kg/m³) is the density of the pore fluid, assumed as 1000 kg/m³, R is the universal gas constant
277 equal to 8.314 [J/(mol K)], T (K) is the test temperature (293.15K) and M (kg/mol) is the molar mass
278 of the pore fluid, assumed equal to that of water, 0.01802 kg/mol.

279 If one considers the different pore pressures acting on the CMIX specimens at equilibrium, the pore
280 pressure in the drying specimens (57% RH) should be always higher than in the specimens in sealed
281 conditions (>80% RH) and hence higher shrinkage should be expected for the drying specimens.
282 However, according to the Biot-Bishop equation (Eq. 1), it would be possible to reach higher
283 autogenous deformation than drying shrinkage if the degree of saturation is substantially different
284 between the two conditions (i.e. higher in the autogenous conditions). It is worth to underline that not
285 only the classical Biot-Bishop approach, but also other poromechanics approaches, e.g. [45–47]
286 would predict higher shrinkage for higher water contents (here expressed with saturation degree)
287 when the pore pressure is the same. The degree of saturation is used to describe the effect of the pore
288 fluid pressure acting only on part of the solid skeleton in unsaturated conditions.

289 In CMIX at initial stages of hydration, a coarse pore structure permits a substantial moisture loss at
290 short drying times [36,48]. In such case, the degree of saturation would drop in drying conditions and,
291 even if the pore fluid pressure is high, the average pore pressure acting on the skeleton could be
292 smaller than in autogenous conditions.

293 In addition, the blended system CMIX shows two distinct hydration phases [36]. For the mortars
294 systems with exactly the same binder as in CMIX, it was concluded that the calcium sulfoaluminate
295 cement reacts immediately after the first addition of water and shows hardly any further reaction after
296 the first week [36]. On the other hand, Portland cement hydration is slower in these systems and does
297 not start until about 3 to 4 weeks after water addition [36]. In autogenous conditions, the internal RH
298 is high enough (about 94% RH was measured in the mortars from 3 to 28 days [36]), to allow
299 hydration of the Portland cement. Cement hydration then induces self-desiccation (the RH decreased
300 to 86% at 56 days [36]) and autogenous shrinkage. The onset of hydration of the Portland cement
301 occurring within the matrix of already hydrated calcium sulfoaluminate cement could also be
302 responsible for the expansion observed between 7 and 28 days (Figure 2c), likely due to
303 crystallization pressure of crystalline hydration products of the Portland cement [49] and possibly due
304 to hygral swelling (note that a slight increase of RH between 3 and 14 days was observed on the
305 corresponding mortar in [36]). On the other hand, in the concrete specimens exposed to drying at
306 early ages, a large amount of moisture is lost, with the consequence that the further reaction of the
307 Portland cement might not occur or occur only to a smaller extent. This would result in no further
308 pore refinement and lower degree of saturation compared to autogenous conditions. While these
309 phenomena would be able to qualitatively explain the observed higher autogenous shrinkage of CMIX
310 compared with its total shrinkage in drying conditions, further research is needed to explain these
311 differences quantitatively. In particular, the different phase composition of the binder in the concrete
312 mixtures (CPC main hydrate C-S-H; CCSA, ettringite and aluminum hydroxide; CMIX ettringite plus

313 C-S-H) and the very different kinetics of hydration of the clinkers may help explaining their shrinkage
314 behavior (see also [36]).

315 The basic creep of CPC is slightly lower compared to CCSA. However, considering the higher
316 strength of CCSA, especially at early ages, and hence the higher applied load, the specific creep of
317 CPC is higher. This can be explained as an effect of: 1) higher mechanical properties (see Fig. 1), and
318 hence possibly also lower creep compliance of the CCSA, and/or 2) by the fact that most of the
319 hydration of the calcium sulfoaluminate cement is over by 1 day of age, when the specimens are
320 exposed to drying, while the Portland cement in CPC would keep reacting for much longer time. The
321 latter explanation is according to a recent model for creep [50], where cement hydration and the
322 resulting dissolution of elastic clinker while the system is subject to external load has been suggested
323 as an important contribution to early-age creep (see also [51]). As regards mechanism 1 (lower
324 specific creep of the CCSA system), it should be also noted that is very likely that the inherent creep
325 properties of the hydration products in the CCSA system are considerably different (here, lower) than
326 in the CPC system. In a microindentation study performed on compacts of hydration products [52],
327 C-S-H had greater creep rate than any other tested hydration product, including ettringite. However,
328 also ettringite exhibited microindentation creep.

329 Regarding the basic creep of the blended system, CMIX is in between the results of the other two
330 systems (considering the applied stresses); the kinetics are clearly different, with gradual increase in
331 basic creep after 2-3 months, while a more pronounced reduction in creep rate could be observed for
332 the other systems. Moreover, the fact that the basic creep of CMIX is higher than its drying creep (the
333 two curves cross at about 70 days), is in line with the free shrinkage results discussed above. This
334 behaviour could be again attributed to long-term, slow hydration of the Portland cement that goes on
335 under external load [50] in the sealed system, while it stops due to insufficient water in the drying
336 system. In the drying system, a moisture gradient and a gradient of degree of hydration (lower degree
337 of hydration close to the surface compared to the centre of the specimen) are expected. The drying
338 front would penetrate more deeply into the sample in the case of slow hydration, i.e. drying would be
339 more important for CMIX compared to CCSA.

340 As regards drying creep, a much higher impact of moisture loss on the creep response could be
341 observed for the CPC system compared to the systems containing CSA binder. Again, this difference
342 may be due to the different hydration products in these systems and hence inherently different creep
343 (both basic and drying) properties.

344 The volume stability, both in terms of shrinkage and creep, is a major concern when dealing with
345 durability of concrete structures. In general, limiting creep is paramount to avoid large deformations
346 of loaded structures, which may lead to failure in extreme cases [53] or to prestress losses [54]. In its
347 turn, shrinkage needs to be limited to avoid that the build-up of restraint stresses causes early-age

348 cracks [55–57], which may reduce the service life of concrete structures. In CPC, the large difference
349 between autogenous and drying shrinkage may induce cracking due to self-restraint in elements with
350 large cross section, since the inner part of the element (in autogenous conditions) shrinks considerably
351 less than the outer layer (in drying conditions). In these terms, both CSA-based systems examined in
352 this study (CCSA and CMIX) would be advantageous, because of their lower shrinkage under drying
353 conditions and because of the smaller difference between autogenous and drying shrinkage. For this
354 latter characteristic, these blended systems are similar to HPC and even more to UHPC [58], in which
355 the contribution to the total shrinkage originating from self-desiccation progressively increases with
356 the decrease of w/c.

357 However, while the long-term shrinkage of CCSA and CMIX does not seem to be of concern,
358 particular attention is required during the very early age. As calcium sulfoaluminate cement is
359 characterized by rapid hydration, self-desiccation and autogenous shrinkage develop at an extremely
360 fast rate already in the first day (evident in a parallel study on mortars [36]), before the start of the
361 length change measurements in this study. Such rapid autogenous shrinkage, which is uniform on the
362 whole cross section of the concrete element, could be counteracted or at least delayed by means of
363 internal curing [59,60]. If self-desiccation and the accompanying shrinkage occur later at a lower rate,
364 there may be sufficient time for stress relaxation (the other manifestation of the viscoelastic properties
365 besides creep) to reduce the stresses so that macroscopic cracking is avoided [61].

366 A final consideration should be made on the creep behaviour of CCSA and CMIX (Figure 3b and c).
367 While for a given applied stress their lower creep compared to CPC would be beneficial, e.g. in
368 limiting the long-term deformation of reinforced concrete structures or prestress losses in prestressed
369 structures, the low creep at early ages may also limit stress relaxation and increase the probability of
370 cracking due to restraint stresses [61,62]. This aspect would require to be investigated in detail in
371 further research. Another important aspect is the characteristic basic creep of CMIX (Figure 3c),
372 which increased steadily even after 6 months of loading.

373 Such a behaviour was even more evident on the measurements made on mortar specimens with the
374 same binders [36]: this could be reasonably linked to the slow reaction of the Portland cement in
375 CMIX when enough water is available in sealed conditions, manifesting as dissolution creep [50]
376 when subjected to mechanical load. This aspect should be investigated in depth before HPC based on
377 blends of PC and CSA cements is used in practical applications as structural concrete.

378

379 **4.2. Comparison with predictions according to Model Code 2010 and ACI 209.R-92**

380 The different regional or national codes contain calculation approaches for both shrinkage and creep
381 of concrete; notable examples are ACI 209.R-92 (1992) [26] and CEB-FIP Model Code 2010 (2010)
382 [25]. Usually, these formulas are based on the knowledge of the 28-days compressive strength of the

383 concrete (or more generally, on the concrete strength class), which is the most widely measured
384 property. In addition, a number of other parameters related to the environmental conditions and the
385 geometry of the concrete member (in particular, to its surface-to-volume ratio) play a role in these
386 formulas.

387 These empirical formulas are based on curve fitting of a large amount of experimental data collected
388 through several decades [63] and almost exclusively regarding normal strength concrete based on
389 Portland cement. Some of these models were recently upgraded to better cover HPC (characterized
390 by rapid hydration and high autogenous shrinkage) and loading at early ages, which is made possible
391 by the very fast strength development [63].

392 In addition, even for relatively small changes between pure Portland cement and blended cements
393 with different types and amounts of supplementary cementitious materials, rather different creep and
394 shrinkage values were measured [22,62]. Given the above, when pure CSA cements or blends of CSA
395 and PC are considered, both their drying behaviour and the deformation as a consequence of self-
396 desiccation and external loads might be radically different from that of concrete of the same strength
397 class based on Portland cement, as discussed in the previous section.

398 In this section, a comparison of the measured creep and shrinkage results (both in sealed and in drying
399 conditions) with the empirical formulas according to Model Code 2010 [25] and ACI 209.R-92 [26]
400 is presented.

401 Considering the creep deformation, the comparison was done in terms of strain (i.e., the total
402 deformation under load after deduction of elastic deformation and shrinkage). In the empirical
403 models, the creep strain was obtained from the creep coefficient prediction, multiplying it by the ratio
404 between the applied stress (which generated the deformation) and the Young's modulus of elasticity
405 at 28 days, calculated from the mean cylindrical compressive strength. The final creep strain is the
406 result of the superposition of three creep coefficient values, as reported in eq. (3), one for each loading
407 step. The first, for age of loading 1 d, was considered throughout the whole investigated period and
408 related to the entire stress applied at 1 d; the second, with age of loading 7 d, was calculated with the
409 incremental stress applied at 7 d (i.e. stress at 7 d minus stress at 1 d); the last, for age of loading 28
410 d, was calculated with the incremental stress applied at 28 d.

$$411 \quad \varepsilon_{cc}(56d) = \frac{\sigma_c(1d)}{E_{28d}} \cdot \varphi(56d, 1d) + \frac{\sigma_c(7d) - \sigma_c(1d)}{E_{28d}} \cdot \varphi(56d, 7d) + \frac{\sigma_c(28d) - \sigma_c(7d)}{E_{28d}} \cdot \varphi(56d, 28d) \quad (3)$$

412 where

413 $\varepsilon_{cc}(t)$ creep strain at time t ;

414 $\sigma_c(t_0)$ applied constant stress at time t_0 ;

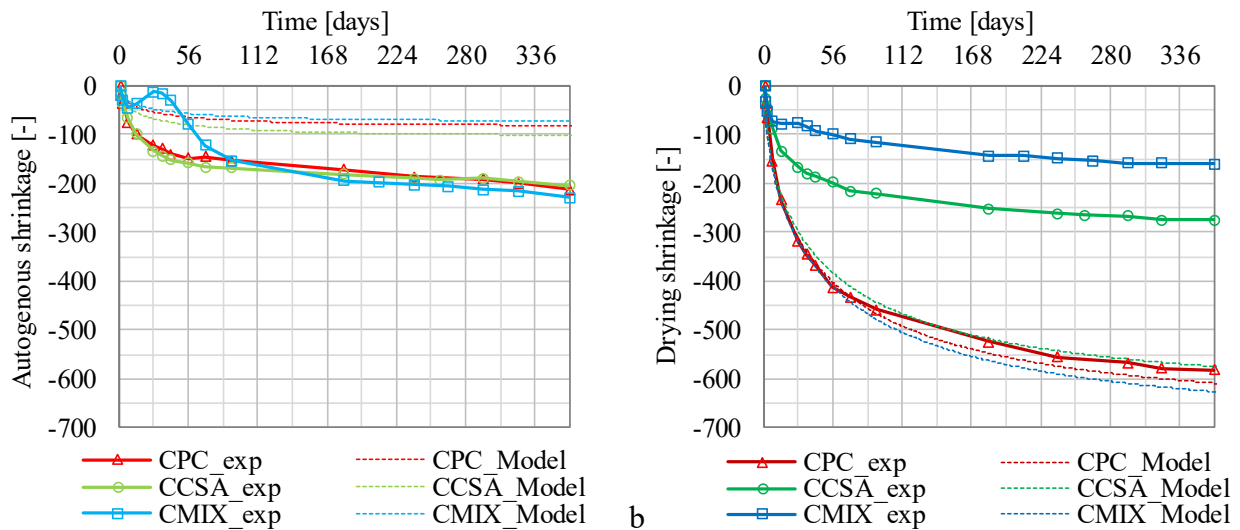
415 E_{ci} dynamic modulus of elasticity at age 28 days;

416 $\varphi(t, t_0)$ creep coefficient at time t , for load applied at time t_0 ;

417 The Model Code 2010 describes both creep and shrinkage as the sum of two parts: one due to
418 autogenous conditions and the other due to drying. These two terms were added when calculating the
419 drying shrinkage and the drying creep. Conversely, ACI 209.R-92 does not distinguish between
420 autogenous and drying conditions; it considers just one formulation describing the total deformation
421 for a structure exposed to drying. As a consequence, ACI 209.R-92 was used only for predicting the
422 total shrinkage in drying conditions and the drying creep. More details about the Model Code 2010
423 and ACI 209.R-92 formulations are reported in Appendix A.

424 Due to their generality and inherent simplifications, it can be expected that the predictive power of
425 these empirical approaches may be limited. In addition, since these formulations are based on the
426 superposition effect, interactions between drying and external loads (i.e. the Pickett effect [64])
427 cannot be taken into account.

428

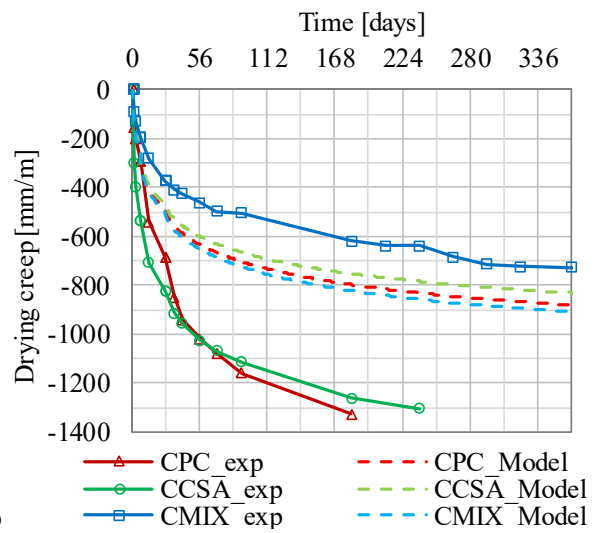
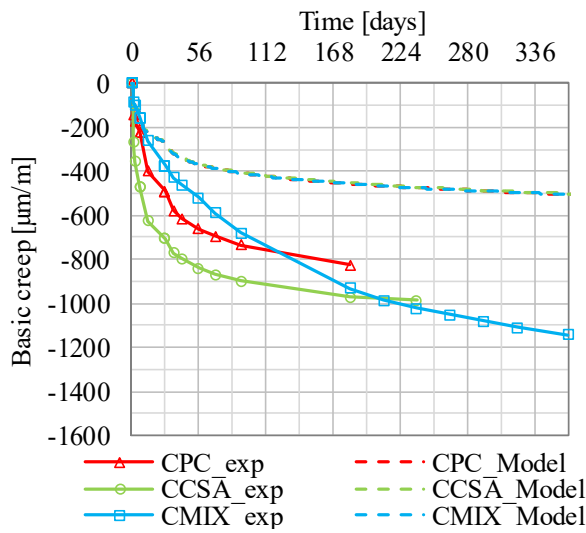


429 a

b

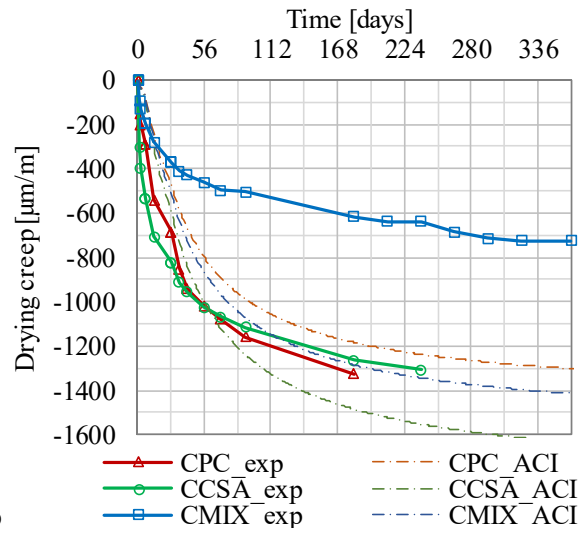
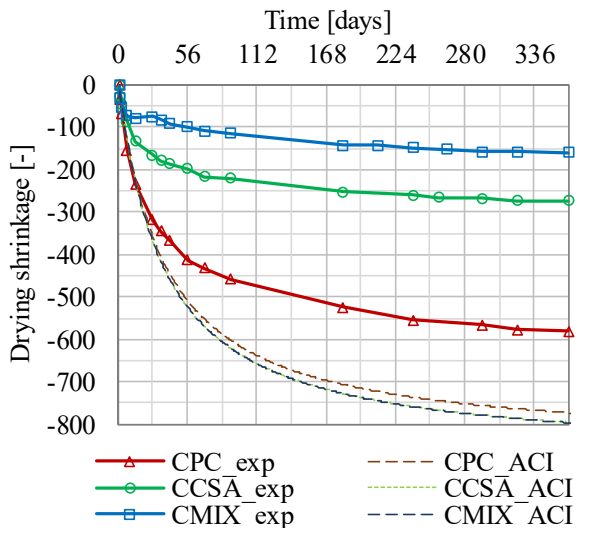
430 *Figure 4. Comparison between experimental and modelled shrinkage evolution in both autogenous (a) and*
431 *drying (b) conditions by Model Code 2010.*

432



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Figure 5. Comparison between experimental and modelled creep evolution in both autogenous (a) and drying (b) conditions by Model code 2010



437
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Figure 6. Comparison between experimental and modelled shrinkage (a) and creep (b) evolution in drying conditions by ACI 209.R-92.

441 Figure 4-6 show the comparison between the predictions with Model Code 2010 and ACI 209.R-92
442 both for creep and shrinkage evolution and the corresponding experiments.

443 Regarding the shrinkage, Model Code 2010 (Figure 4) predicts very well the total shrinkage under
444 drying conditions of the PC system, while the autogenous shrinkage of sealed specimens is
445 underestimated (by more than 50%). Considering that the Model Code 2010 calculates the total
446 shrinkage under drying conditions as the sum of an autogenous shrinkage contribution and a drying
447 shrinkage contribution (see Appendix A), these findings lead to the conclusion that the drying
448 shrinkage contribution to the total shrinkage in drying conditions tends to be overestimated. In the
449 cases of CCSA and CMIX, Model Code 2010 underestimates the shrinkage in sealed conditions

450 similarly as for CPC, while it overestimates considerably the shrinkage in drying conditions. While
451 for CCSA the predicted shrinkage is more than twice the experimental value, the overestimation for
452 CMIX is more than fourfold. On the other hand, ACI 209.R-92 (Figure 6a) overestimates the ultimate
453 shrinkage in drying conditions by about 15%; while the shape of the curves is similar; after 28 days
454 the predicted deformation increases faster than the experimental.

455 Compared to ACI 209.R-92, Model Code 2010 predicts more accurately the short term deformation
456 after the load increments. However, for both autogenous and drying conditions, the long-term
457 deformation is underestimated for the PC system. On the contrary, the ACI model is less accurate in
458 the short term after loading but predicts ultimate values close to those obtained experimentally for
459 the PC system.

460 The predictions of basic creep according to Model Code 2010 (Figure 5a) underestimate considerably
461 the experimental results for all three concrete mixtures (the predictions are about half of the
462 experimental values). The case of drying creep predictions with Model Code 2010 (Figure 5b) is
463 similar for the CPC and CCSA, while for CMIX the predictions are about 35% higher. On the
464 contrary, ACI 209.R-92 (only for drying creep) seems to be more precise when predicting drying
465 creep of CCSA and CPC (10% and 20% difference, respectively, see Figure 6b), while it
466 overestimates considerably the drying creep of CMIX.

467 When only CPC is considered, it appears that both models give reasonably good predictions in terms
468 of drying shrinkage; furthermore ACI 209.R-92 seems reliable for drying creep as well. The
469 autogenous shrinkage and basic creep models in Model Code 2010 appear to underestimate the
470 deformations considerably.

471 Given the above, while the accuracy of these models cannot obviously be judged based only on few
472 experimental results, the differences in sealed conditions are rather large and may be worth a deeper
473 investigation at these models in further research.

474 For the systems based on calcium sulfoaluminate cement (CCSA and CMIX), it can be concluded
475 that the experimental results are considerably different from the predictions of both models.
476 Nevertheless, under drying conditions, the shape of the experimental and predicted curves is similar
477 and the order of magnitude is the same. Thus, it is expected that special factors can be defined for the
478 equations used in both Model Code 2010 and ACI 209.R-92 (see Appendix A) in order to take into
479 account the initial rapid and intense self-desiccation and autogenous shrinkage of CCSA and the
480 smaller overall deformation of CMIX that were observed under all conditions. Finally, for CMIX
481 made with blended cement, the autogenous deformation showed a succession of shrinkage and
482 expansion phases that would require entirely different models. Similarly, the fact that the basic creep
483 of CMIX was larger than the drying creep would require an *ad-hoc* description different from the
484 current approach in Model Code 2010.

485 If CSA cement and blends with Portland cement are to be included in practical models as Model Code
486 2010 and ACI 209.R-92, a large basis of experimental data covering multiple concrete mixtures,
487 geometries, exposure and loading conditions will be required.

488 In the meantime, based on projects like the one described in this paper, a database with measurements
489 of mechanical properties, deformation behaviour and durability indicators of structural concrete based
490 on blends of CSA and Portland cement will be established. This will help concrete producers,
491 engineers and contractors in the choice of appropriate mixture compositions for the different practical
492 applications until mature engineering models for concrete with these novel binders become available.

493

494 **5. CONCLUSIONS**

495 In this research, the shrinkage and creep evolution in both autogenous and drying conditions of HPC
496 based on CSA cement and blends of CSA with Portland cement were studied.

497 Three concrete mixtures based on different cement systems were investigated: a Portland-limestone
498 cement (CEM II-LL 42.5R), a commercial CSA and a blend of the two previous cements at ratio of
499 50/50 by mass.

500 Both the creep and the shrinkage of CSA-based systems were lower than for the PC system. The
501 concrete with pure CSA cement showed rapid and significant self-desiccation and autogenous
502 shrinkage, while the additional shrinkage in drying conditions was limited. CSA-based mixtures
503 showed also lower differences between basic and drying creep. On the contrary, CPC reacts slowly
504 and the differences between autogenous shrinkage and drying shrinkage, and between basic and
505 drying creep are significant. The blended system CMIX showed the lowest deformation in both
506 shrinkage and creep when exposed to drying, with limited differences between drying and sealed
507 conditions. Interestingly, both shrinkage and creep in sealed conditions were slightly higher than in
508 drying conditions, which may be explained by continued hydration, pore refinement and higher
509 degree of saturation of the sealed systems.

510 The measured creep and shrinkage of these concrete mixtures were then compared to the empirical
511 models according to Model Code 2010 and ACI 209.R-92, which were developed based on Portland
512 cement. While the predictions of the deformation of the PC concrete under drying conditions were
513 satisfactory, Model Code 2010 underestimated the deformation in sealed conditions (ACI 209.R-92
514 does not cover this case). For CCSA and CMIX, different empirical factors should be defined in the
515 models so as to take into account the initial strong self-desiccation of CCSA and the global lower
516 shrinkage of CMIX. On the other hand, previous studies found that current empirical models are
517 effective in predicting compressive strength and modulus of elasticity of these mixtures. While the
518 definition of special models for the deformation behaviour (shrinkage and creep) of CSA cements in
519 these codes would need a substantial basis of experimental work, the results presented in this paper,

520 together with other ongoing campaigns, can be an initial guidance for engineers and contractors that
521 are working with CSA cement.

522

523 6. REFERENCES

- 524 [1] M.C.G. Juenger, F. Winnefeld, J.L. Provis, J.H. Ideker, Advances in alternative cementitious
525 binders, *Cem. Concr. Res.* 41 (2011) 1232–1243. doi:10.1016/j.cemconres.2010.11.012.
- 526 [2] F.P. Glasser, L. Zhang, High-performance cement matrices based on calcium sulfoaluminate-
527 belite compositions, *Cem. Concr. Res.* 31 (2001) 1881–1886. doi:10.1016/S0008-
528 8846(01)00649-4.
- 529 [3] F. Winnefeld, B. Lothenbach, Hydration of calcium sulfoaluminate cements - Experimental
530 findings and thermodynamic modelling, *Cem. Concr. Res.* 40 (2010) 1239–1247.
531 doi:10.1016/j.cemconres.2009.08.014.
- 532 [4] J. Péra, J. Ambroise, E. Holard, G. Beauvent, Influence of the type of calcium sulfate on the
533 properties of calcium sulfoaluminate cement, in: 11th Int. Congr. Chem. Cem., 2003: pp.
534 1129–1135.
- 535 [5] L. Pelletier, F. Winnefeld, B. Lothenbach, The ternary system Portland cement-calcium
536 sulphoaluminate clinker-anhydrite: Hydration mechanism and mortar properties, *Cem. Concr.*
537 *Compos.* 32 (2010) 497–507. doi:10.1016/j.cemconcomp.2010.03.010.
- 538 [6] W. Lan, F.P. Glasser, Hydration of calcium sulphoaluminate cements, *Adv. Cem. Res.* 8
539 (1996) 127–134.
- 540 [7] L. Zhang, F.P. Glasser, Hydration of calcium sulfoaluminate cement at less than 24 h, *Adv.*
541 *Cem. Res.* 14 (2002) 141–155. doi:10.1680/adcr.2002.14.4.141.
- 542 [8] D. Sirtoli, S. Tortelli, P. Riva, M. Marchi, R. Cucitore, M.N. Rose, Mechanical and durability
543 performances of sulfo-based rapid hardening concrete, *ACI Spec. Publ.* 305 (2015) 1–8.
- 544 [9] L. Zhang, M. Su, Y. Wang, Development of the use of sulfo- and ferroaluminate cements in
545 China, *Adv. Cem. Res.* 11 (1999) 15–21. doi:10.1680/adcr.1999.11.1.15.
- 546 [10] M.S. Meddah, M. Suzuki, R. Sato, Influence of a combination of expansive and shrinkage-
547 reducing admixture on autogenous deformation and self-stress of silica fume high-performance
548 concrete, *Constr. Build. Mater.* 25 (2011) 239–250. doi:10.1016/j.conbuildmat.2010.06.033.
- 549 [11] S. Slatnick, K.A. Riding, K.J. Folliard, M.C.G. Juenger, A.K. Schindler, Evaluation of
550 autogenous deformation of concrete at early ages, *ACI Mater. J.* 108 (2011) 21–28.
551 doi:10.14359/51664212.
- 552 [12] T. Noguchi, P. Sun-Gyu, I. Maruyama, Mechanical properties of high-performance concrete
553 with expansive additive and shrinkage reducing admixtures under simulated completely-
554 restrained condition at early age, in: *Self-Desiccation Its Importance Concr. Technol. Proc. 4th*

- 555 Int. Semin., 2005.
- 556 [13] J. Péra, J. Ambroise, New applications of calcium sulfoaluminate cement, *Cem. Concr. Res.*
557 34 (2004) 671–676. doi:10.1016/j.cemconres.2003.10.019.
- 558 [14] J. Ambroise, J. Péra, Use of calcium sulfoaluminate cement to improve strength of mortars at
559 low temperature, *Concr. Repair, Rehabil. Retrofitting II.* (2009) 881–886.
- 560 [15] L. Pelletier-Chaignat, F. Winnefeld, B. Lothenbach, C.J. Müller, Beneficial use of limestone
561 filler with calcium sulphoaluminate cement, *Constr. Build. Mater.* 26 (2012) 619–627.
562 doi:10.1016/j.conbuildmat.2011.06.065.
- 563 [16] I.A. Chen, C.W. Hargis, M.C.G. Juenger, Understanding expansion in calcium sulfoaluminate-
564 belite cements, *Cem. Concr. Res.* 42 (2012) 51–60. doi:10.1016/j.cemconres.2011.07.010.
- 565 [17] J. Weiss, P. Lura, F. Rajabipour, G. Sant, Performance of Shrinkage-Reducing Admixtures at
566 Different Humidities and at Early Ages, *ACI Mater. J.* 105 (2008). doi:10.14359/19977.
- 567 [18] O. Mejlhede Jensen, P. Freiesleben Hansen, Autogenous deformation and change of the
568 relative humidity in silica fume-modified cement paste, *ACI Mater. J.* 93 (1996) 539–543.
569 doi:10.14359/9859.
- 570 [19] P. Lura, O.M. Jensen, K. Van Breugel, Autogenous shrinkage in high-performance cement
571 paste: An evaluation of basic mechanisms, *Cem. Concr. Res.* 33 (2003) 223–232.
572 doi:10.1016/S0008-8846(02)00890-6.
- 573 [20] P. Lura, F. Winnefeld, X. Fang, A simple method for determining the total amount of
574 physically and chemically bound water of different cements, *J. Therm. Anal. Calorim.* (2015)
575 1–8.
- 576 [21] S.I. Igarashi, A. Bentur, K. Kovler, Autogenous shrinkage and induced restraining stresses in
577 high-strength concretes, *Cem. Concr. Res.* 30 (2000) 1701–1707. doi:10.1016/S0008-
578 8846(00)00399-9.
- 579 [22] P. Lura, K. Van Breugel, I. Maruyama, Effect of curing temperature and type of cement on
580 early-age shrinkage of high-performance concrete, *Cem. Concr. Res.* 31 (2001) 1867–1872.
581 doi:10.1016/S0008-8846(01)00601-9.
- 582 [23] D. Sirtoli, P. Riva, S. Tortelli, M. Marchi, Mechanical performance of sulfo-based rapid
583 hardening concrete, in: *New Boundaries Struct. Concr. Sess. C - New Scenar. Concr.*, 2016.
- 584 [24] D. Sirtoli, P. Riva, S. Tortelli, M. Marchi, Mechanical Performance Comparison Between
585 Sulfo-Based and Portland Concretes, *Spec. Publ.* (2018).
- 586 [25] J. Walraven, Model Code 2010, final drafts, *FIB Bull.* 1 & 2 (2012) 105.
- 587 [26] ACI Committee 209, ACI 209R-92 Prediction of Creep, Shrinkage, and Temperature Effects
588 in Concrete Structures, 2008. doi:10.1201/9780203882955.ch168.
- 589 [27] L.H.J. Martin, F. Winnefeld, C.J. Müller, B. Lothenbach, Contribution of limestone to the

- 590 hydration of calcium sulfoaluminate cement, *Cem. Concr. Compos.* 62 (2015) 204–211.
591 doi:10.1016/j.cemconcomp.2015.07.005.
- 592 [28] T. Matschei, B. Lothenbach, The role of calcium carbonate in cement hydration, *Cem. Concr.*
593 *Res.* 37 (2007) 551–558.
- 594 [29] Z.P. Bazant, Mathematical modelling of creep and shrinkage of concrete, *Mathematical Model.*
595 *Creep Shrinkage Concr.* (1988) 99–215.
- 596 [30] H.M. Jennings, Colloid model of C-S-H and implications to the problem of creep and
597 shrinkage, *Mater. Struct.* (2003). doi:10.1617/14137.
- 598 [31] L.G. Baquerizo, T. Matschei, K.L. Scrivener, Impact of water activity on the stability of
599 ettringite, *Cem. Concr. Res.* (2016). doi:10.1016/j.cemconres.2015.07.008.
- 600 [32] J. Kaufmann, F. Winnefeld, B. Lothenbach, Stability of ettringite in CSA cement at elevated
601 temperatures, *Adv. Cem. Res.* (2016). doi:10.1680/jadcr.15.00029.
- 602 [33] J. Bizzozero, C. Gosselin, K.L. Scrivener, Expansion mechanisms in calcium aluminate and
603 sulfoaluminate systems with calcium sulfate, *Cem. Concr. Res.* 56 (2014) 190–202.
604 doi:10.1016/j.cemconres.2013.11.011.
- 605 [34] P. Chaunsali, P. Mondal, Influence of Calcium Sulfoaluminate (CSA) Cement Content on
606 Expansion and Hydration Behavior of Various Ordinary Portland Cement-CSA Blends, *J. Am.*
607 *Ceram. Soc.* 8 (2015) n/a-n/a. doi:10.1111/jace.13645.
- 608 [35] P. Chaunsali, P. Mondal, Physico-chemical interaction between mineral admixtures and OPC-
609 calcium sulfoaluminate (CSA) cements and its influence on early-age expansion, *Cem. Concr.*
610 *Res.* 80 (2016) 10–20. doi:10.1016/j.cemconres.2015.11.003.
- 611 [36] D. Sirtoli, M. Wyrzykowski, F. Winnefeld, P. Riva, P. Lura, Volume changes of mortars based
612 on calcium sulfoaluminate cement, Submitted to *Cement and Concrete Research*, 2019.
- 613 [37] M. Wyrzykowski, P. Lura, Effect of relative humidity decrease due to self-desiccation on the
614 hydration kinetics of cement, *Cem. Concr. Res.* 85 (2016) 75–81.
615 doi:10.1016/j.cemconres.2016.04.003.
- 616 [38] T.C. Powers, A Discussion of Cement Hydration in Relation to the Curing of Concrete, *Highw.*
617 *Res. Board Proc.* 27 (1948) 178–188.
- 618 [39] O.M. Jensen, Thermodynamic limitation of self-desiccation, *Cem. Concr. Res.* 25 (1995) 157–
619 164. doi:10.1016/0008-8846(94)00123-G.
- 620 [40] R.J. Flatt, G.W. Scherer, J.W. Bullard, Why alite stops hydrating below 80% relative humidity,
621 *Cem. Concr. Res.* 41 (2011) 987–992. doi:10.1016/j.cemconres.2011.06.001.
- 622 [41] O. Coussy, *Poromechanics*, 2004. doi:10.1002/0470092718.
- 623 [42] H. Chen, M. Wyrzykowski, K. Scrivener, P. Lura, Prediction of self-desiccation in low water-
624 to-cement ratio pastes based on pore structure evolution, *Cem. Concr. Res.* 49 (2013) 38–47.

- 625 doi:10.1016/j.cemconres.2013.03.013.
- 626 [43] Z. Hu, M. Wyrzykowski, K. Scrivener, P. Lura, A novel method to predict internal relative
627 humidity in cementitious materials by ^1H NMR, *Cem. Concr. Res.* (2017).
628 doi:10.1016/j.cemconres.2017.11.001.
- 629 [44] P. Lura, B. Lothenbach, Influence of Pore Solution Chemistry on Shrinkage of Cement Paste,
630 50-Year Teach. Res. Anniv. Prof. Sun Wei Adv. Civ. Eng. Mater. (2010) 191–200.
- 631 [45] I. Vlahinić, H.M. Jennings, J.J. Thomas, A constitutive model for drying of a partially saturated
632 porous material, *Mech. Mater.* 41 (2009) 319–328. doi:10.1016/j.mechmat.2008.10.011.
- 633 [46] O. Coussy, R. Eymard, T. Lassabatère, Constitutive Modeling of Unsaturated Drying
634 Deformable Materials, *J. Eng. Mech.* 124 (1998) 658–667. doi:10.1061/(ASCE)0733-
635 9399(1998)124:6(658).
- 636 [47] J.-M. Pereira, O. Coussy, E.E. Alonso, J. Vaunat, S. Olivella, Is the degree of saturation a good
637 candidate for Bishop's X parameter?, Alonso, E.E., Gens, A. Unsaturated Soils—Proceedings
638 5th Int. Conf. Unsaturated Soils. (2010) 913–919. doi:10.1016/j.cemconres.2010.12.004.
- 639 [48] D. Sirtoli, S. Tortelli, M. Marchi, P. Lura, Drying and Autogenous Shrinkage Evolution of a
640 Blended CSA-Portland Cement Concrete, *Spec. Publ.* 326 (2018).
- 641 [49] G. Sant, B. Lothenbach, P. Juilland, G. Le Saout, J. Weiss, K. Scrivener, The origin of early
642 age expansions induced in cementitious materials containing shrinkage reducing admixtures,
643 *Cem. Concr. Res.* (2011). doi:10.1016/j.cemconres.2010.12.004.
- 644 [50] X. Li, Z.C. Grasley, E.J. Garboczi, J.W. Bullard, Modeling the apparent and intrinsic
645 viscoelastic relaxation of hydrating cement paste, *Cem. Concr. Compos.* 55 (2015) 322–330.
646 doi:10.1016/j.cemconcomp.2014.09.012.
- 647 [51] M. Wyrzykowski, K. Scrivener, P. Lura, Basic creep of cement paste at early age—the role of
648 cement hydration, *Cem. Concr. Res.* 116 (2019) 191–201.
- 649 [52] D.T. Nguyen, R. Alizadeh, J.J. Beaudoin, L. Raki, Microindentation creep of secondary
650 hydrated cement phases and C-S-H, *Mater. Struct. Constr.* (2013). doi:10.1617/s11527-012-
651 9993-0.
- 652 [53] Z.P. Bažant, M.H. Hubler, Q. Yu, Pervasiveness of excessive segmental bridge deflections:
653 Wake-up call for creep, *ACI Struct. J.* 108 (2011) 766–774.
654 doi:10.1017/CBO9781107415324.004.
- 655 [54] T.D. Lämmlein, F. Messina, M. Wyrzykowski, G.P. Terrasi, P. Lura, Low clinker high
656 performance concretes and their potential in CFRP prestressed structural elements, *Cem.*
657 *Concr. Res.* (2018).
- 658 [55] N. Neithalath, B. Pease, J.H. Moon, F. Rajabipour, J. Weiss, E. Attiogbe, Considering moisture
659 gradient and time-dependent crack growth in restrained concrete elements subjected to drying,

- 660 NSF Work. High Perform. Concr. (2005) 279–290.
- 661 [56] V. Gribniak, G. Kaklauskas, R. Kliukas, R. Jakubovskis, Shrinkage effect on short-term
662 deformation behavior of reinforced concrete – When it should not be neglected, Mater. Des.
663 51 (2013) 1060–1070. doi:10.1016/j.matdes.2013.05.028.
- 664 [57] A. Bentur, Evaluation of early age cracking characteristics in cementitious systems, Mater.
665 Struct. 36 (2003) 183–190. doi:10.1617/14014.
- 666 [58] P.-C. Aïtcin, High performance concrete, CRC press, 2011.
- 667 [59] O.M. Jensen, P. Lura, Techniques and materials for internal water curing of concrete, Mater.
668 Struct. 39 (2006) 817–825. doi:10.1617/s11527-006-9136-6.
- 669 [60] P. Lura, O.M. Jensen, S.I. Igarashi, Experimental observation of internal water curing of
670 concrete, Mater. Struct. Constr. 40 (2007) 211–220. doi:10.1617/s11527-006-9132-x.
- 671 [61] A.B. Hossain, J. Weiss, Assessing residual stress development and stress relaxation in
672 restrained concrete ring specimens, Cem. Concr. Compos. 26 (2004) 531–540.
673 doi:10.1016/S0958-9465(03)00069-6.
- 674 [62] A. Leemann, P. Lura, R. Loser, Shrinkage and creep of SCC - The influence of paste volume
675 and binder composition, Constr. Build. Mater. 25 (2011) 2283–2289.
676 doi:10.1016/j.conbuildmat.2010.11.019.
- 677 [63] H.S. Müller, I. Anders, R. Breiner, M. Vogel, Concrete: Treatment of types and properties in
678 fib Model Code 2010, Struct. Concr. 14 (2013) 320–334. doi:10.1002/suco.201200048.
- 679 [64] G. Pickett, The effect of change in moisture-content on the creep of concrete under a sustained
680 load, J. Am. Concr. Inst. 13 (1942) 333–355. doi:10.14359/8607.

681

682 **Appendix A – Technical documents formulation**

683 For defining the shrinkage and creep functions according to the codes, the following aspects were
684 considered. The measured average cubic compressive strength of each concrete was multiplied by
685 0.83 to transform it into an average cylindrical compressive strength f_{cm} . Other aspects defined for
686 the model are the applied stress, which is one third of the compressive strength at the considered time
687 of loading (25% for CPC), the relative humidity, which is that of the testing room (57% RH), the time
688 of loading t_0 , which takes into account the different loading steps (see section 2.5) and is adjusted by
689 the room temperature of 20°C, the cement type, where a rapid hardening CEM I 42.5R is taken for
690 all concrete mixtures (since no CSA cement is considered in the code), and the concrete age at the
691 beginning of drying, set to 1 day.

692 A cement type III was considered in the ACI 209.R-92 model, as CEM I 42.5R is not included in that
693 model. Moreover, only for the ACI 209.R-92 model, f and α for the shrinkage prediction and d and
694 Ψ for the creep prediction were taken as constant for a given member shape and size (α and Ψ equal

695 to 1 for a flatter hyperbolic curve), all the composition parameters were referred to test values on
696 fresh state just before the casting operation (slump, air content, volumetric mass) and as “fine
697 aggregate” in the fine aggregate factor calculation the fraction below 3 mm was considered.
698 In the ACI 209.R-92 model, the creep was obtained in terms of creep coefficient. The subsequent
699 extrapolation of the creep strain was obtained following the Model Code 2010 formulation in order
700 to have similar references. The ACI 209.R-92 model considers as modulus of elasticity the one
701 obtained according to the code based on the 28-days strength.

702

703 Model Code 2010 - Creep strain

704
$$\varepsilon_{cc}(t, t_0) = \frac{\sigma_c(t_0)}{E_{ci}} \cdot \varphi(t, t_0)$$

705 with

706
$$E_{ci} = E_{c0} \cdot \alpha_E \cdot \left(\frac{f_{cm}}{10}\right)^{1/3}$$

707
$$\varphi(t, t_0) = \varphi_{bc}(t, t_0) + \varphi_{dc}(t, t_0)$$

708 where

709 $\varepsilon_{cc}(t, t_0)$ creep strain at time $t > t_0$;

710 $\sigma_c(t_0)$ applied constant stress at time t_0 ;

711 E_{ci} dynamic modulus of elasticity at age 28 days;

712 $\varphi(t, t_0)$ creep coefficient;

713 E_{c0} assumed equal to $21.5 \cdot 10^3$;

714 α_E constant, assumed to be 1 for quartzite aggregates;

715 f_{cm} mean compressive strength at age 28 days;

716 $\varphi_{bc}(t, t_0)$ basic creep coefficient;

717 $\varphi_{dc}(t, t_0)$ drying creep coefficient;

718

719 Model Code 2010 - Basic creep coefficient

720
$$\varphi_{bc}(t, t_0) = \beta_{bc}(f_{cm}) \cdot \beta_{bc}(t, t_0)$$

721 with

722
$$\beta_{bc}(f_{cm}) = \frac{1.8}{(f_{cm})^{0.7}}$$

723
$$\beta_{bc}(t, t_0) = \ln \left[\left(\frac{30}{t_{0,adj}} + 0.035 \right)^2 \cdot (t - t_0) + 1 \right]$$

724
$$t_{0,adj} = t_{0,T} \cdot \left[\frac{9}{2+t_{0,T}^{1.2}} + 1 \right]^\alpha \geq 0.5 \text{ days}$$

725
$$t_{0,T} = \sum_{i=1}^n \Delta t_i \exp \left[13.65 - \frac{4000}{273+T(\Delta t_i)} \right]$$

726 where

727 $t_{0,adj}$ modified age of loading, taking into account the type of cement and the temperature;

728 α coefficient that depends on the type of cement;

729 $t_{0,T}$ temperature-adjusted concrete age;

730 Δt_i number of days where a temperature T prevails;

731 $T(\Delta t_i)$ temperature in °C during the time period Δt_i .

732

733 Model Code 2010 - Drying creep coefficient

734
$$\varphi_{dc}(t, t_0) = \beta_{dc}(f_{cm}) \cdot \beta(RH) \cdot \beta_{dc}(t_0) \cdot \beta_{dc}(t, t_0)$$

735 with

736
$$\beta_{dc}(f_{cm}) = \frac{412}{(f_{cm})^{1.4}}$$

737
$$\beta(RH) = \frac{1 - \frac{RH}{100}}{\sqrt[3]{0.1 \cdot \frac{h}{100}}}$$

738
$$h = \frac{2 \cdot A_C}{u}$$

739
$$\beta_{dc}(t_0) = \frac{1}{0.1 + (t_{0,adj})^{0.2}}$$

740
$$\beta_{dc}(t, t_0) = \left[\frac{(t - t_0)}{\beta_h + (t - t_0)} \right]^{\gamma(t_0)}$$

741
$$\gamma(t_0) = \frac{1}{2.3 + \frac{3.5}{\sqrt{t_{0,adj}}}}$$

742
$$\beta_h = 1.5 \cdot h + 250 \cdot \alpha_{f_{cm}} \leq 1500 \cdot \alpha_{f_{cm}}$$

743
$$\alpha_{f_{cm}} = \left(\frac{35}{f_{cm}} \right)^{0.5}$$

744 where

745 RH relative humidity of the ambient environment in %;

746 h notional size of member, where A_C is the cross-section in mm² and u is the perimeter of
747 the member in contact with the atmosphere in mm.

748

749 Model Code 2010 - Shrinkage

750
$$\varepsilon_{cs}(t, t_s) = \varepsilon_{cas}(t) + \varepsilon_{cds}(t, t_s)$$

751 with

752 $\varepsilon_{cas}(t) = \varepsilon_{cas0}(f_{cm}) \cdot \beta_{as}(t)$

753 $\varepsilon_{cas0}(f_{cm}) = -\alpha_{as} \left(\frac{f_{cm}/10}{6+f_{cm}/10} \right)^{2.5} \cdot 10^{-6}$

754 $\beta_{as}(t) = 1 - \exp(-0.2 \cdot \sqrt{t})$

755 $\varepsilon_{cds}(t, t_s) = \varepsilon_{cds0}(f_{cm}) \cdot \beta_{RH}(RH) \cdot \beta_{ds}(t - t_s)$

756 $\varepsilon_{cds0}(f_{cm}) = [(220 + 110 \cdot \alpha_{ds1}) \cdot \exp(-\alpha_{ds2} \cdot f_{cm})] \cdot 10^{-6}$

757 $\beta_{RH}(RH) = f(x) = \begin{cases} -1.55 \cdot \left[1 - \left(\frac{RH}{100} \right)^3 \right], & \text{for } 40 \leq RH < 99\% \cdot \beta_{s1} \\ 0.25, & \text{for } RH \geq 99\% \cdot \beta_{s1} \end{cases}$

758 $\beta_{s1} = \left(\frac{35}{f_{cm}} \right)^{0.1} \leq 1.0$

759 $\beta_{ds}(t - t_s) = \left[\frac{(t-t_s)}{0.035 \cdot h^2 + (t-t_s)} \right]^{0.5}$

760 where

761 $\varepsilon_{cas}(t)$ autogenous shrinkage;

762 α_{as} coefficient dependent on the type of cement;

763 $\varepsilon_{cds}(t, t_s)$ drying shrinkage;

764 $\alpha_{ds1}, \alpha_{ds2}$ coefficients dependent on the type of cement;

765 t_s concrete age at the beginning of drying, in days.

766

767 ACI 209.R-92 - Creep strain

768 $\varepsilon_{cc}(t, t_0) = \frac{\sigma_c(t_0)}{E_{cmto}} \cdot \varphi(t, t_0)$

769 with

770 $E_{cmto} = 0.043 \cdot \gamma_c^{1.5} \cdot \sqrt{f_{cmto}}$

771 $f_{cmto} = \left[\frac{t}{a+bt} \right] \cdot f_{cm28}$

772 $\varphi(t, t_0) = \frac{(t - t_0)^\Psi}{d + (t - t_0)^\Psi} \cdot \varphi_u$

773 $d = 4 \cdot \frac{V}{S}$

774 $\varphi_u = 2.35 \cdot \gamma_c$

775 $\gamma_c = \gamma_{c,t0} \cdot \gamma_{c,RH} \cdot \gamma_{c,vs} \cdot \gamma_{c,s} \cdot \gamma_{c,\Psi} \cdot \gamma_{c,\alpha}$

776 $\gamma_{c,t0} = 1.25 \cdot t_0^{-0.118}$ for moist curing

777 $\gamma_{c,RH} = 1.27 - 0.67 \cdot h$ for $h \geq 0.40$

778 $\gamma_{c,vs} = \frac{2}{3} \cdot (1 + 1.13 \cdot e^{\{-0.0213(V/S)\}})$ in SI units

779 $\gamma_{c,s} = 0.82 + 0.00264 \cdot s$ in SI units

780 $\gamma_{c,\Psi} = 0.88 + 0.0024 \cdot \Psi$

781 $\gamma_{c,\alpha} = 0.46 + 0.09 \cdot \alpha \geq 1$

782 where

783 $\varepsilon_{cc}(t, t_0)$ creep strain at time $t > t_0$;

784 $\sigma_c(t_0)$ applied constant stress at time t_0 ;

785 E_{cmt_0} modulus of elasticity at time t_0 ;

786 $\varphi(t, t_0)$ creep coefficient;

787 γ_c unit weight of concrete;

788 f_{cmt_0} mean compressive strength at time t_0 ;

789 a, b constants referred to a moist cured type III cement;

790 d, Ψ constant for a given member shape and size;

791 φ_u ultimate creep coefficient;

792 γ_c cumulative product of the applicable correction factors;

793 γ_{c,t_0} age of loading factor;

794 $\gamma_{c,RH}$ ambient relative humidity factor;

795 $\gamma_{c,vs}$ size of member factor;

796 $\gamma_{c,s}$ slump factor;

797 $\gamma_{c,\Psi}$ fine aggregate factor;

798 $\gamma_{c,\alpha}$ air content factor.

799

800 ACI 209.R-92 - Shrinkage

801
$$\varepsilon_{sh}(t, t_c) = \frac{(t - t_c)^\alpha}{f + (t - t_c)^\alpha} \cdot \varepsilon_{shu}$$

802
$$f = 26 \cdot e^{\{1.42 \times 10^{-2} (\frac{V}{S})\}}$$
 in SI units

803
$$\varepsilon_{shu} = 780 \cdot \gamma_{sh} \times 10^{-6} [mm/mm]$$

804
$$\gamma_{sh} = \gamma_{sh,t_c} \cdot \gamma_{sh,RH} \cdot \gamma_{sh,vs} \cdot \gamma_{sh,s} \cdot \gamma_{sh,\Psi} \cdot \gamma_{sh,c} \cdot \gamma_{sh,\alpha}$$

805
$$\gamma_{sh,t_c} = 1.202 - 0.2337 \cdot \log(t_c)$$

806
$$\gamma_{sh,RH} = \begin{cases} 1.40 - 1.02 \cdot h, & \text{for } 0.40 \leq h < 0.80 \\ 3.00 - 3.0 \cdot h, & \text{for } 0.80 \leq h < 1 \end{cases}$$

807
$$\gamma_{sh,vs} = 1.2 \cdot e^{\{-0.00472(V/S)\}}$$
 in SI units

808
$$\gamma_{sh,s} = 0.89 + 0.00161 \cdot s$$
 in SI units

809
$$\gamma_{sh,\Psi} = \begin{cases} 0.30 + 0.014 \cdot \Psi, & \text{for } \Psi \leq 50\% \\ 0.90 + 0.002 \cdot \Psi, & \text{for } \Psi > 50\% \end{cases}$$

810
$$\gamma_{sh,c} = 0.75 + 0.00061 \cdot c$$
 in SI units

811
$$\gamma_{sh,\alpha} = 0.95 + 0.008 \cdot \alpha \geq 1$$

812 where

813 $\varepsilon_{sh}(t, t_c)$ shrinkage strain at time $t > t_c$;

814 f, α constants depending on member shape and size;

815 ε_{shu} ultimate shrinkage strain;

816 γ_{sh} cumulative product of the applicable correction factors;

817 $\gamma_{sh,t0}$ age of loading factor;

818 $\gamma_{sh,RH}$ ambient relative humidity factor;

819 $\gamma_{sh,vs}$ size of the member factor;

820 $\gamma_{sh,s}$ slump factor;

821 $\gamma_{sh,\Psi}$ fine aggregate factor;

822 $\gamma_{sh,c}$ cement content factor;

823 $\gamma_{sh,\alpha}$ air content factor.